

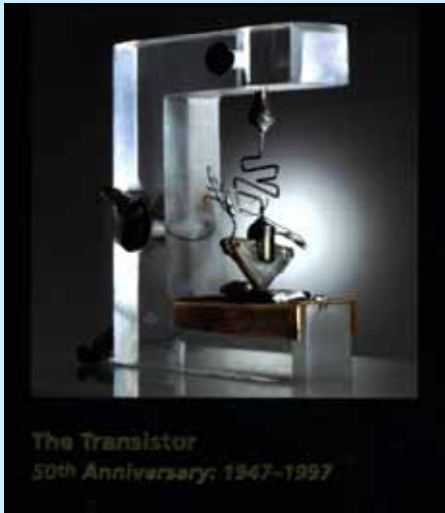
Nanoelectronics Beyond Si: Challenges and Opportunities

Prof. J. Raynien Kwo 郭瑞年
國立清華大學

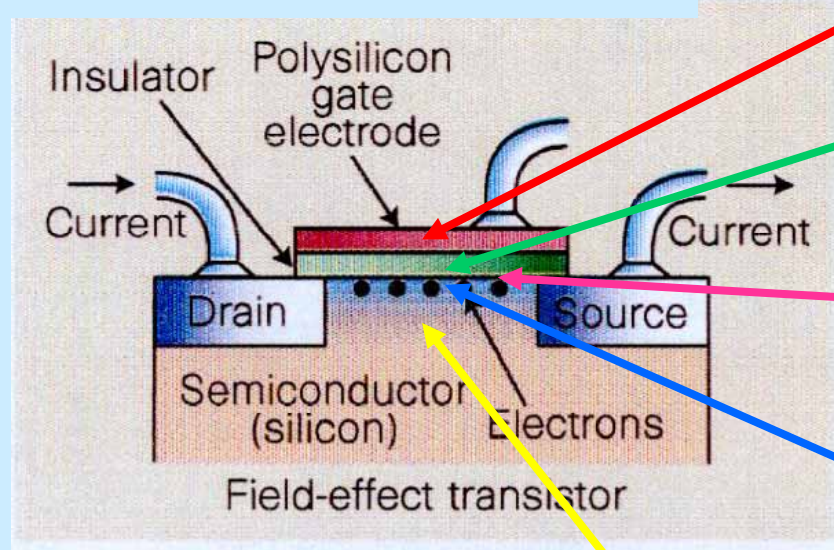
Si CMOS Device Scaling – Beyond 22 nm node

High κ , Metal gates, and High mobility channel

1947 First Transistor



1960 First MOSFET



Metal Gate

High κ gate dielectric

Oxide/semiconductor interface

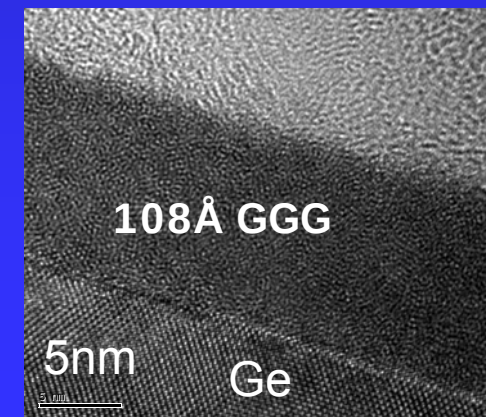
High mobility channel

Moore's Law: The number of transistors per square inch doubles every 18 months

Integration of Ge, III-V with Si

Shorter gate length L
Thinner gate dielectrics t_{ox}

Driving force :
High speed
Low power consumption
High package density



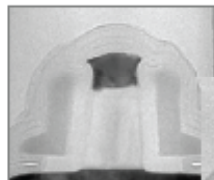
Intel Transistor Scaling and Research Roadmap

Ultimate scaling of CMOS

(22 nm node and beyond)

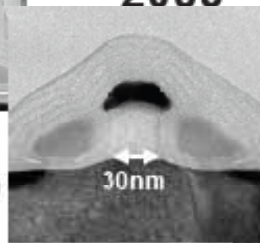
More non-silicon elements introduced

90nm Node
2003



(Production)

65nm Node
P1264
2005

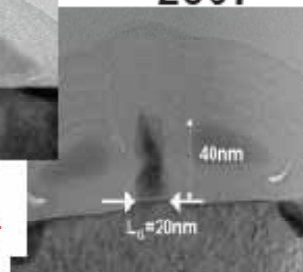


(Development)
(production)

Uniaxial
Strain

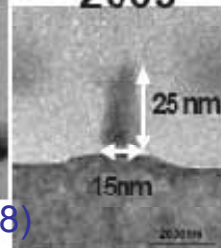
SiGe S/D

45nm Node
P1266
2007



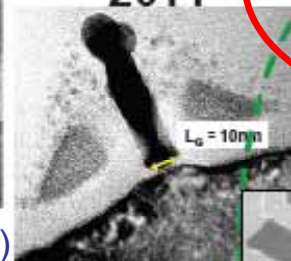
(Development)
(production-early2008)

32nm Node
P1268
2009



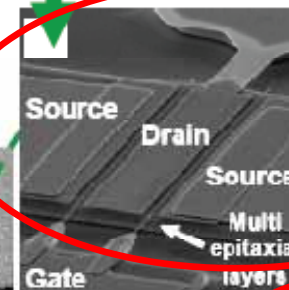
(development)

22nm Node
P1270
2011



(Research)

2013-2019



III-V Device
Prototype
(Research)

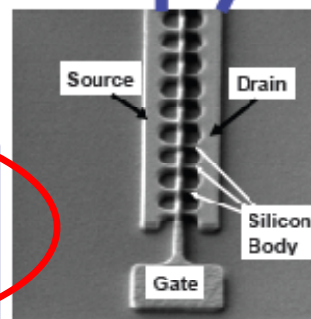


C-nanotube
Prototype
(Research)



Ge Device
Prototype
(Research)

High-K/
Metal-Gate



Tri-Gate
Architecture

Major Research Subjects

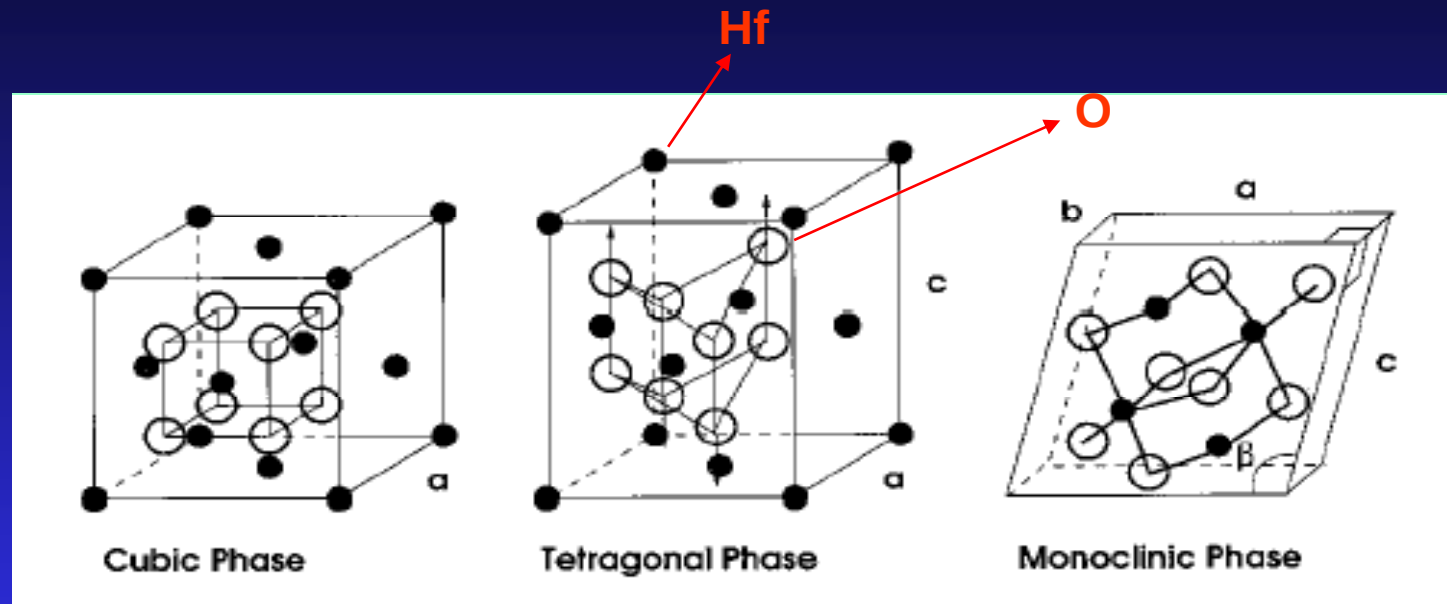
- Enhancement of κ in the new phase through epitaxy
- Fundamental study by IETS for detections of phonons and defects in high κ dielectrics
- Room temperature ferromagnetism in cluster free, Co doped HfO_2 films

Can you make κ even higher ?

“ Phase Transition Engineering ”

---Enhancement of κ in the New Phase
through Epitaxy

Crystal structures of HfO_2 and the corresponding κ



Dielectric constants 29 70 16 *

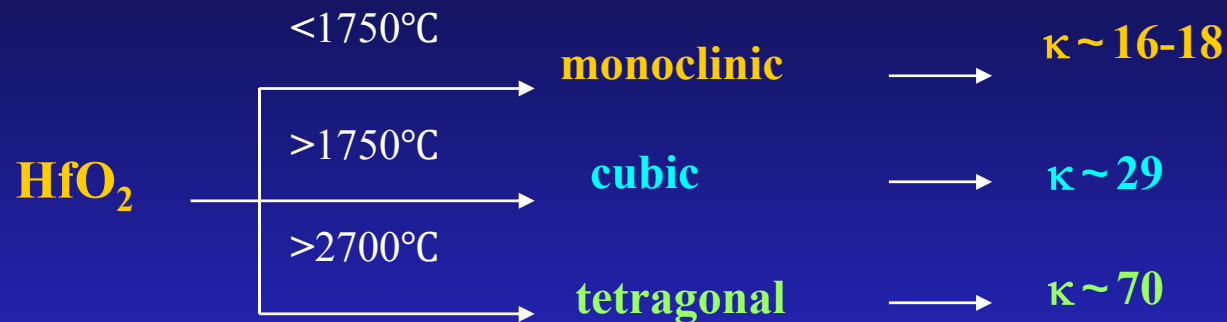
Stable phase temperature $>2700^\circ\text{C}$ $>1750^\circ\text{C}$ $<1750^\circ\text{C}$

The dielectric constant increases when HfO_2 structure is changed from monoclinic to other symmetry

* Xinyuan Zhao and David Vanderbit, P.R.B. 65, 233106, (2002).

Permittivity Increase of Yttrium-doped HfO₂ Through Structural Phase Transformation

by Koji Kita, Kentaro Kyuno, and Akira Toriumi, Tokyo Univ.



- ❖ Yttrium serves effectively as a dopant to induce a phase transformation from the *monoclinic* to the *cubic* phase even at 600 °C.
- ❖ Yttrium-doped HfO₂ films show higher permittivity than undoped HfO₂, and the permittivity as high as 27 is obtained by 4 at. % yttrium doping.
- ❖ The permittivity of undoped HfO₂ is reduced significantly at high temperature, whereas that of 17 at. % yttrium-doped film shows no change even at 1000 °C.

After deposition
for 5mins

Dielectric film of
4-fold symmetry
in the plane

After deposition
for 2mins

After
reconstruction

2x position

1x position

190°

280°

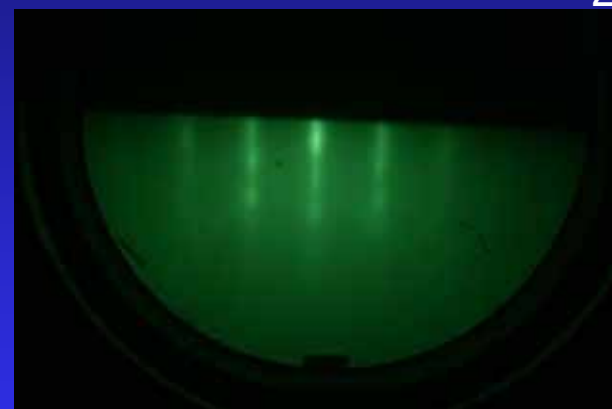
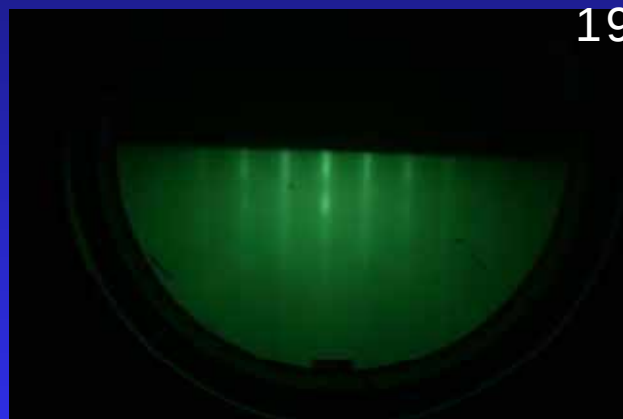
190°

280°

145°

Wafer rotate 22.5°

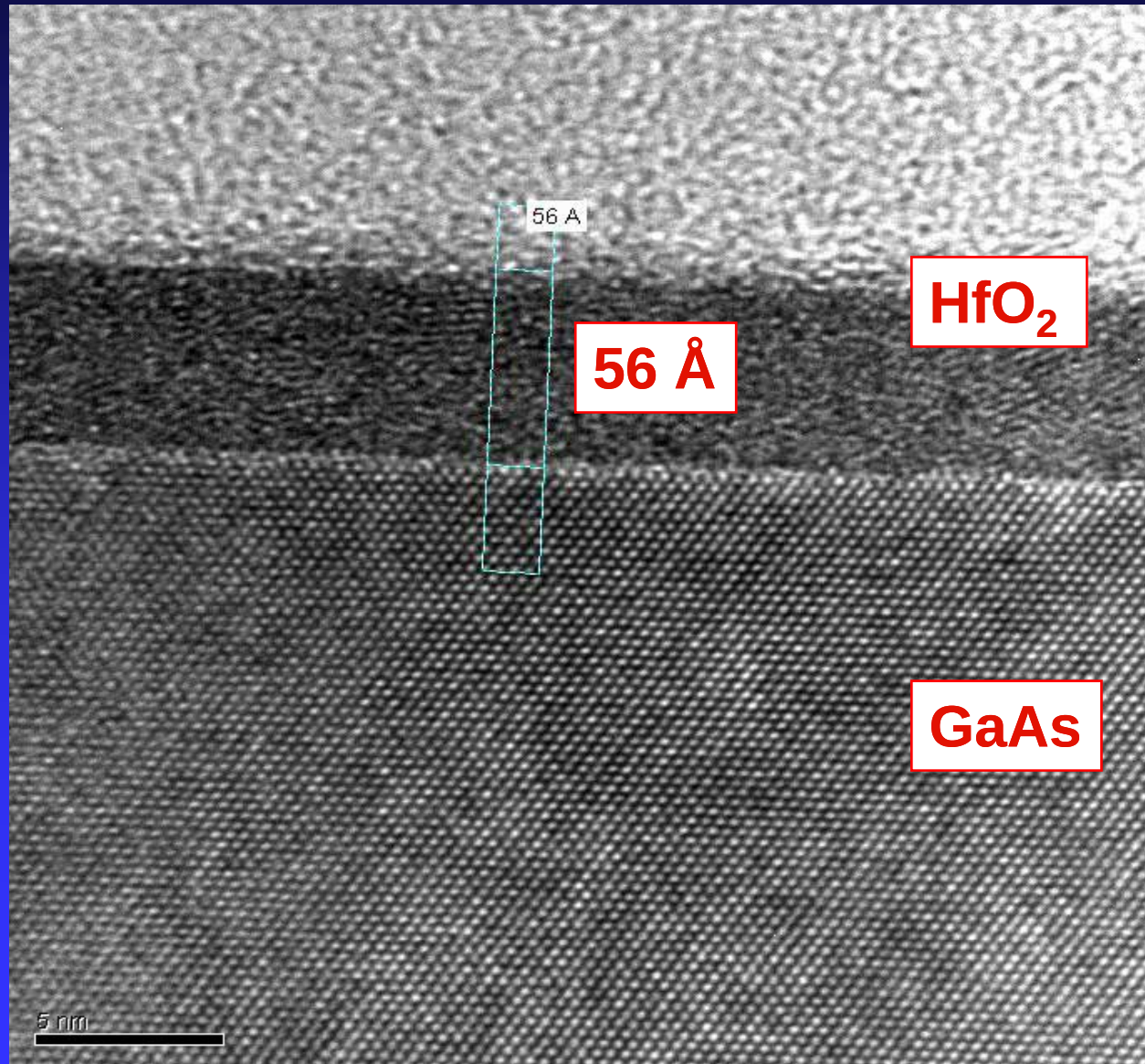
235°



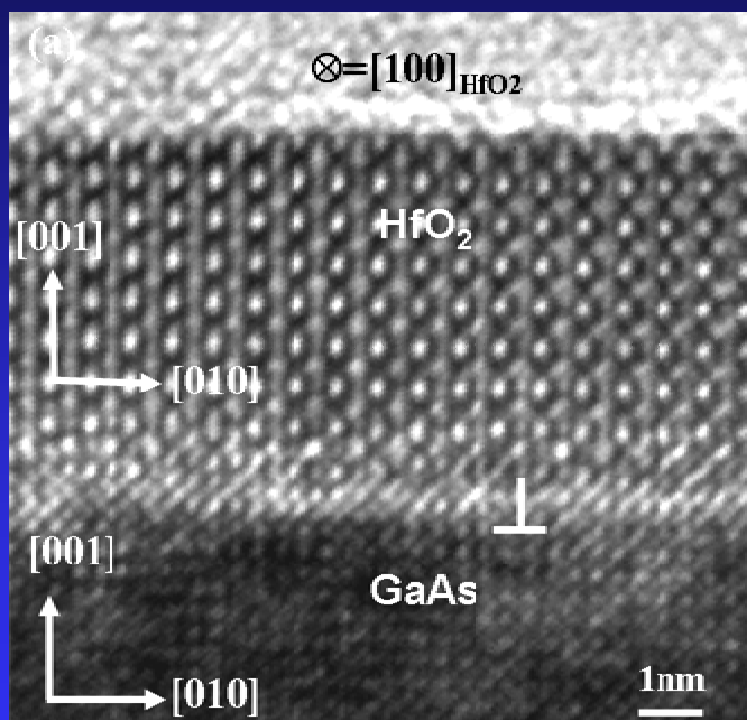
HRTEM of Low Temp Growth

Amorphous HfO_2
on GaAs (100)

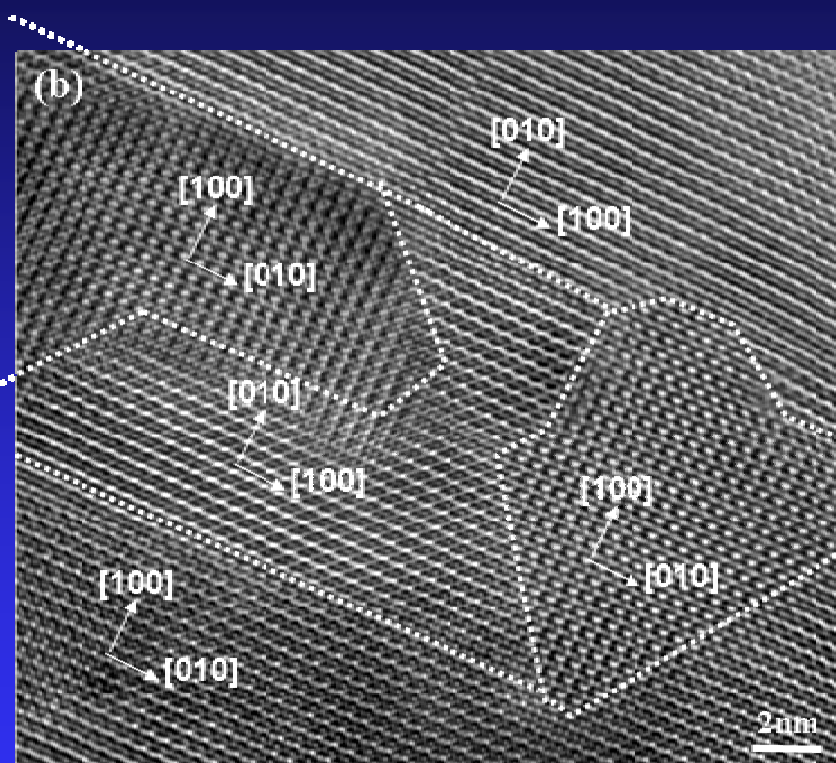
A very abrupt
transition from
GaAs to HfO_2 over
one atomic layer
thickness was
observed.



High Resolution TEM Images of Pure HfO_2 on GaAs (001)

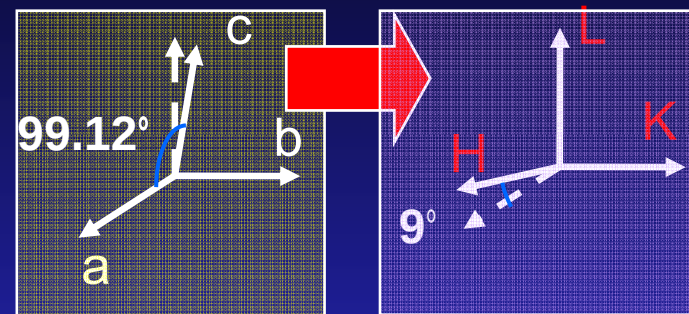
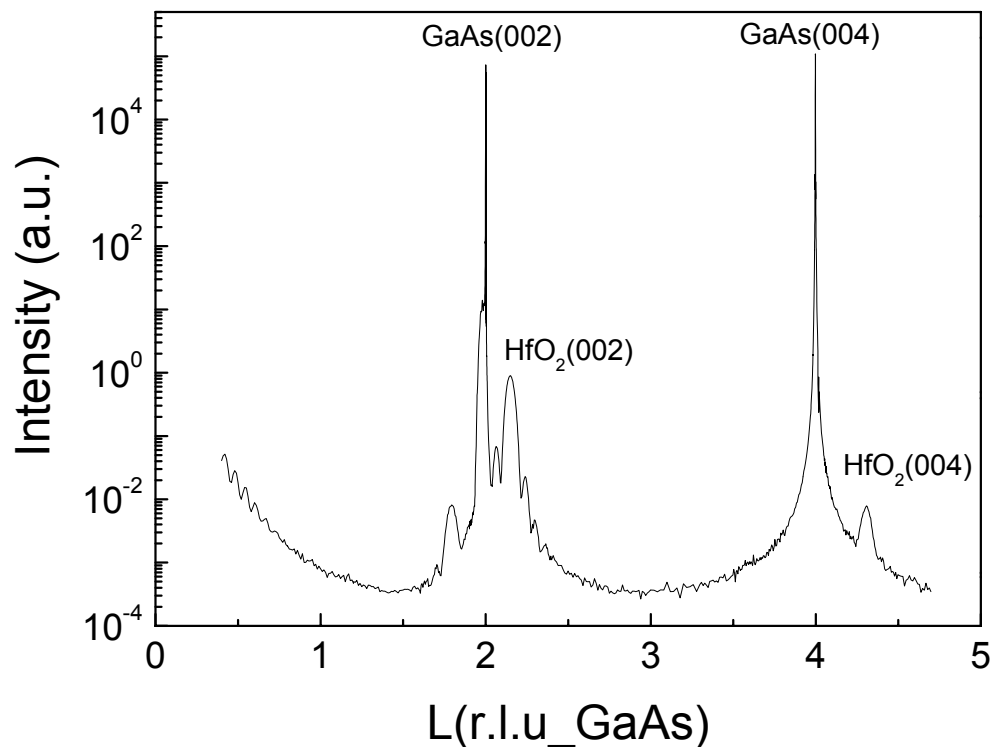


An abrupt transition from GaAs to HfO_2 and no interfacial layer



Coexistence of four monoclinic domains rotated by 90° .

X-ray Diffraction of Epitaxial HfO₂ Films Recrystallized on GaAs



R space

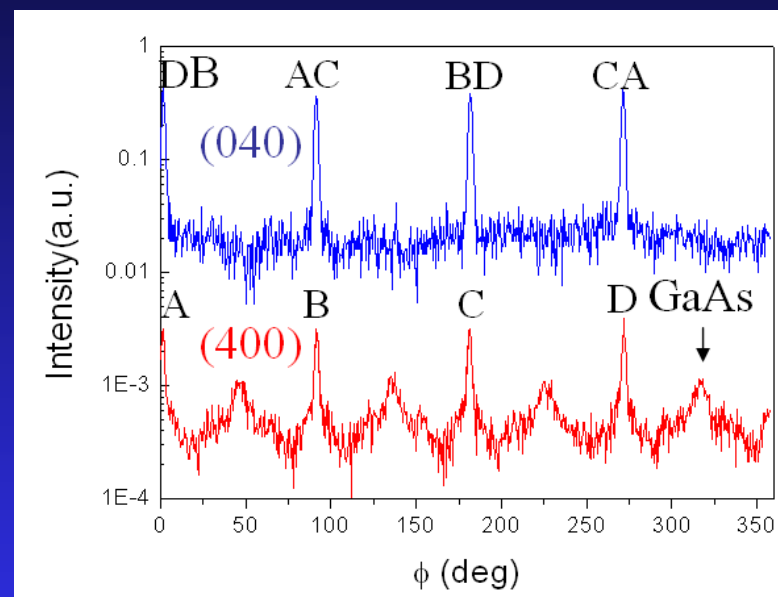
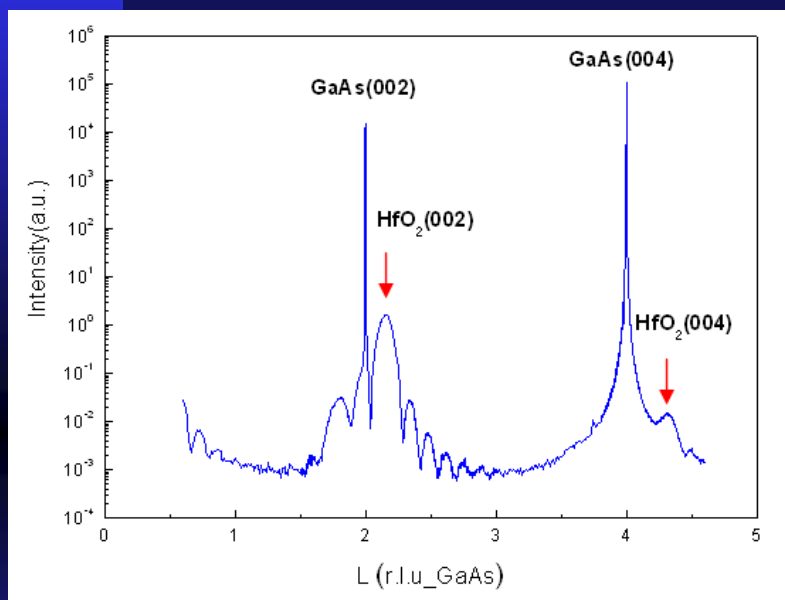
K space

--- Monoclinic HfO₂ in R space and K space
--- Forming four degenerate domains about the surface normal

With C. H. Hsu of NSRRC

HfO₂(004) FWHM(L)=0.0578°
⇒ domain size 97.8 Å
⇒ close to film thickness

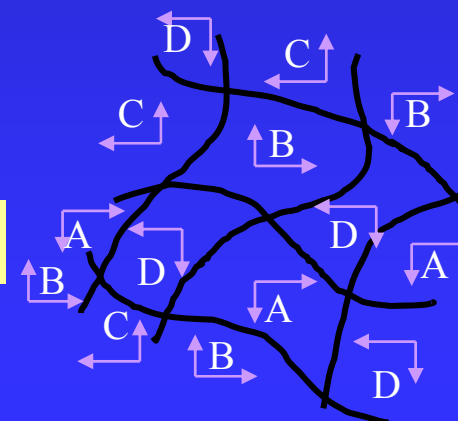
The Structure of HfO_2 Grown on GaAs(001)



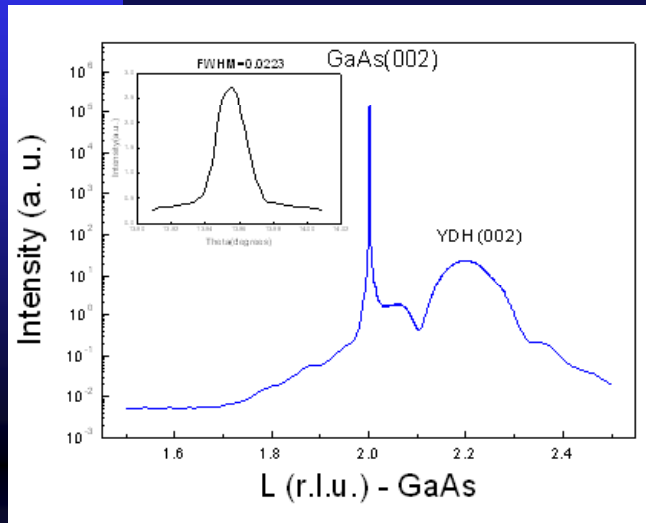
monoclinic phase

$a=5.116\text{\AA}$, $b=5.172\text{\AA}$, $c=5.295\text{\AA}$, $\beta=99.18^\circ$

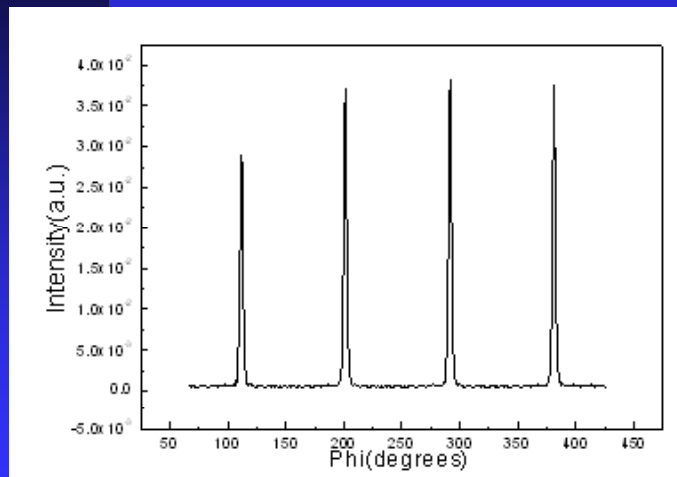
Coexistence of 4 domains rotated 90° from each other



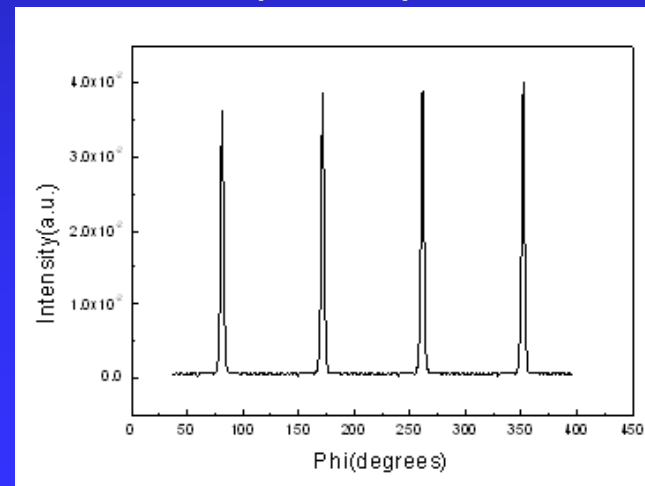
The Structure of HfO_2 doped with Y_2O_3 Grown on GaAs (100)



Surface normal scan



(400)Phi scan



(040)Phi scan

Find the peaks:

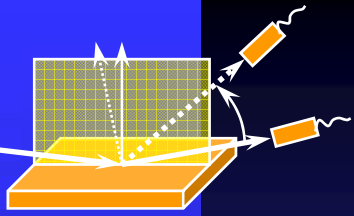
- (022)(400)(200)(311)(31-1)(113)(420)(133)(20-2)
- All peaks of film match the JCPDS of cubic phase HfO_2
- Use the d-spacing formula to fit the lattice parameters
→ HfO_2 doped with Y_2O_3 Grown on GaAs(001) is

Cubic phase

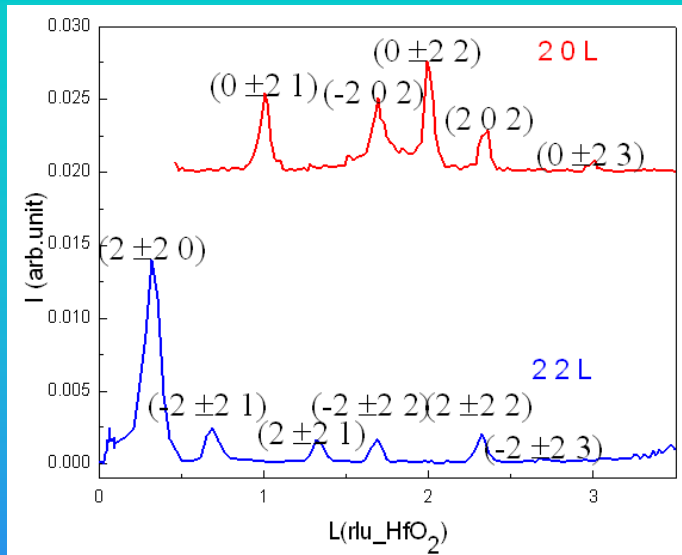
$a=5.126\text{\AA}$, $b=5.126\text{\AA}$, $c=5.126\text{\AA}$

$\alpha=90^\circ$, $\beta=90^\circ$, $\gamma=90^\circ$

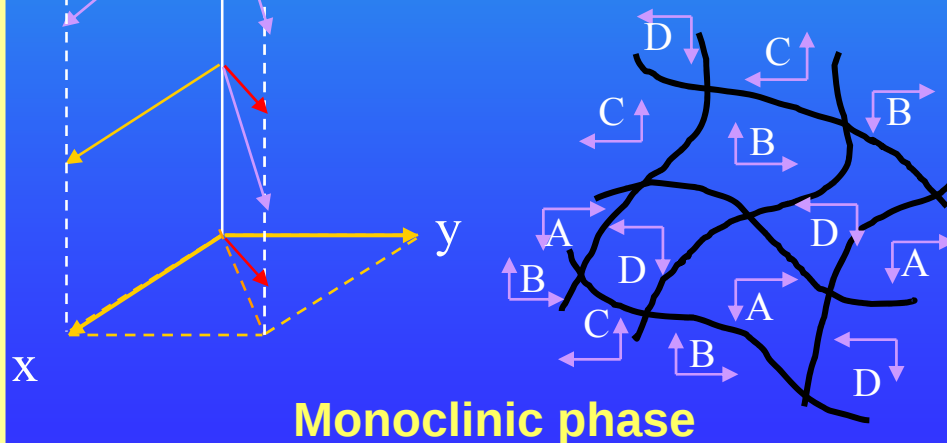
Comparison between Monoclinic phase and Cubic phase of HfO_2



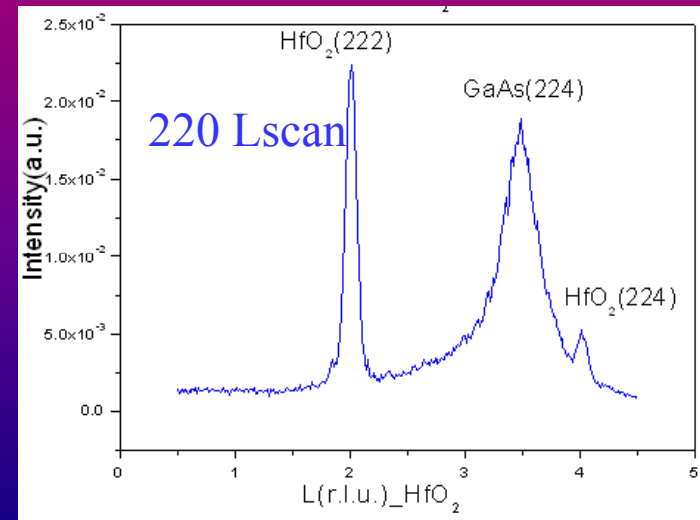
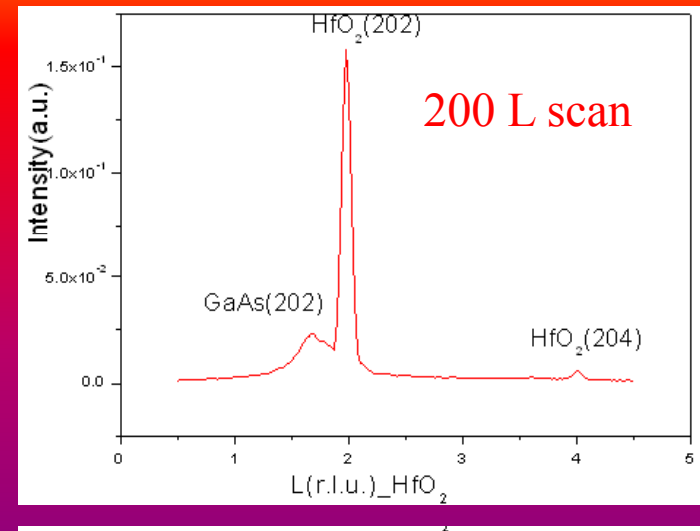
Without doping Y_2O_3



200 & 220 L scan

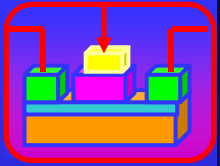


With doping Y_2O_3

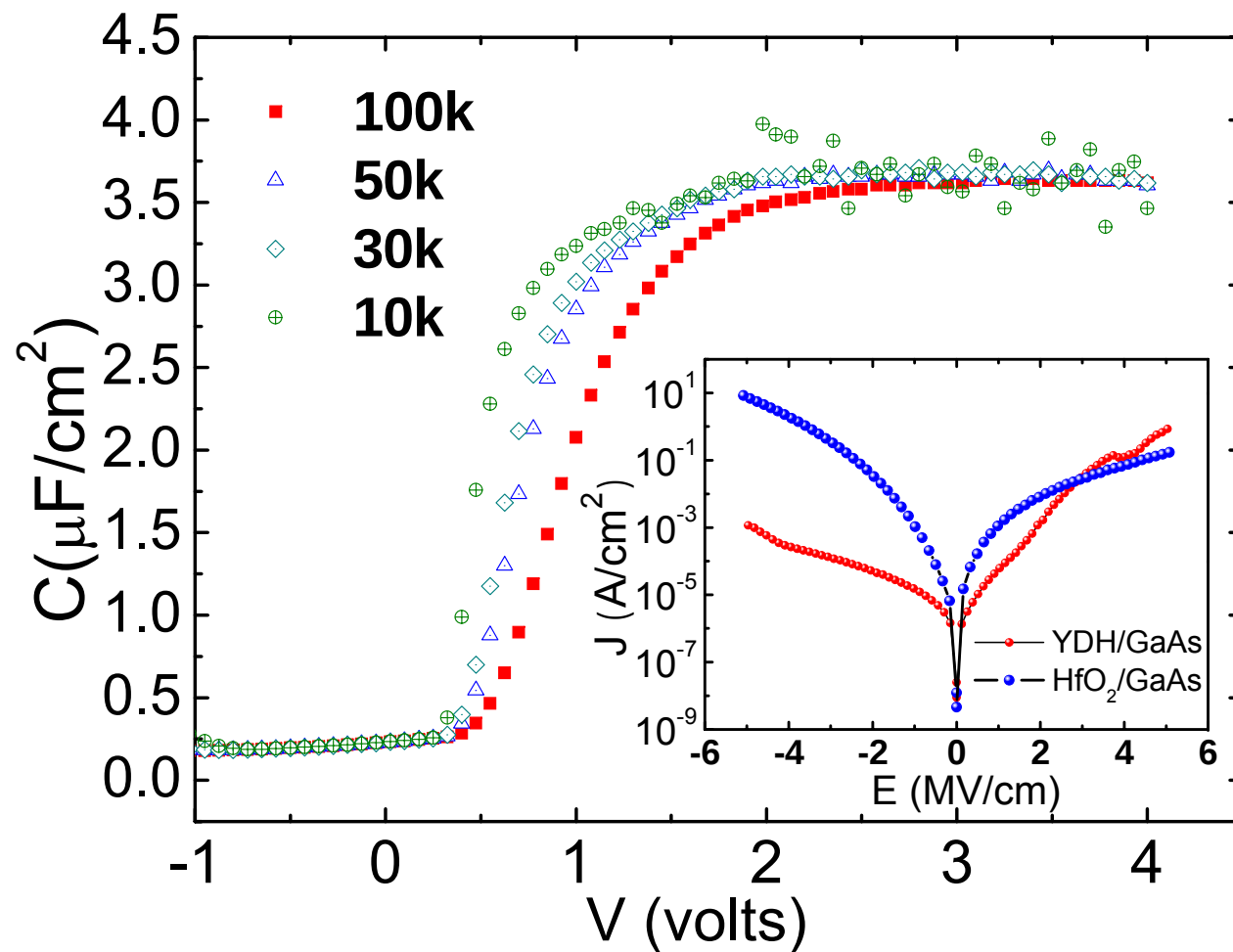
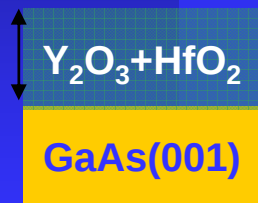


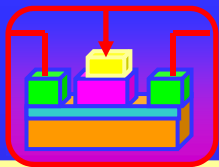
Cubic phase

The Electrical Property of HfO_2 doped with Y_2O_3 Grown on $\text{GaAs}(001)$



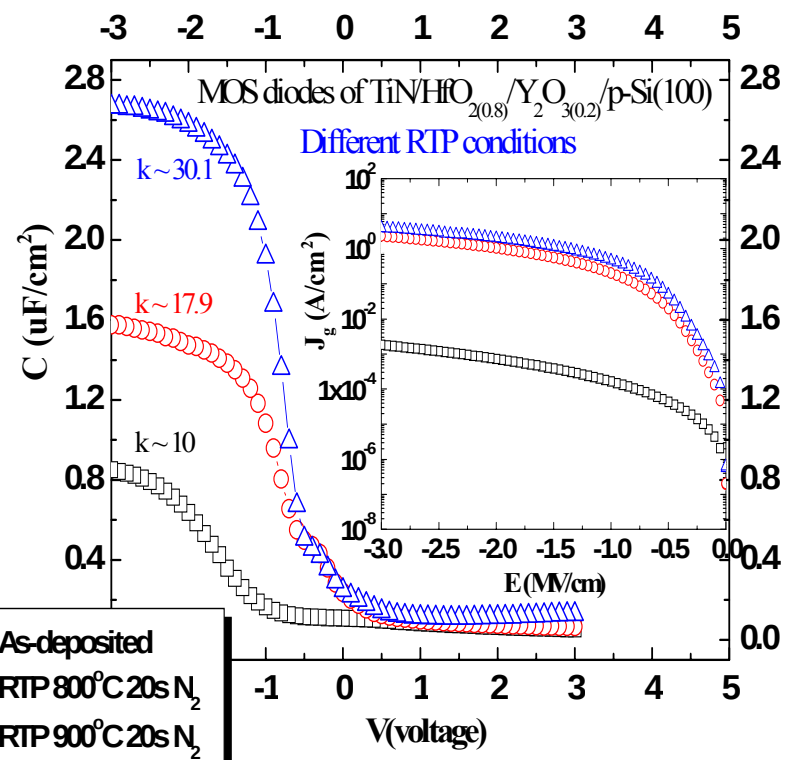
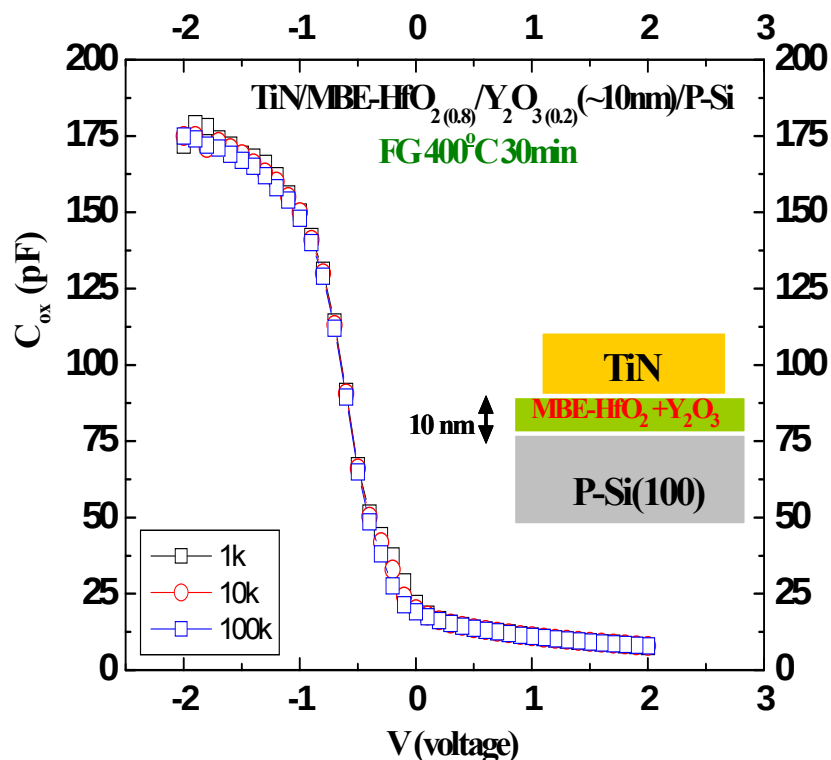
$T=110\text{A}$ $\kappa=32$





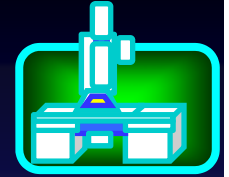
Electrical properties - MBE-HfO₂(0.8)/Y₂O₃(0.2)

MBE-HfO₂(0.8)/Y₂O₃(0.2)

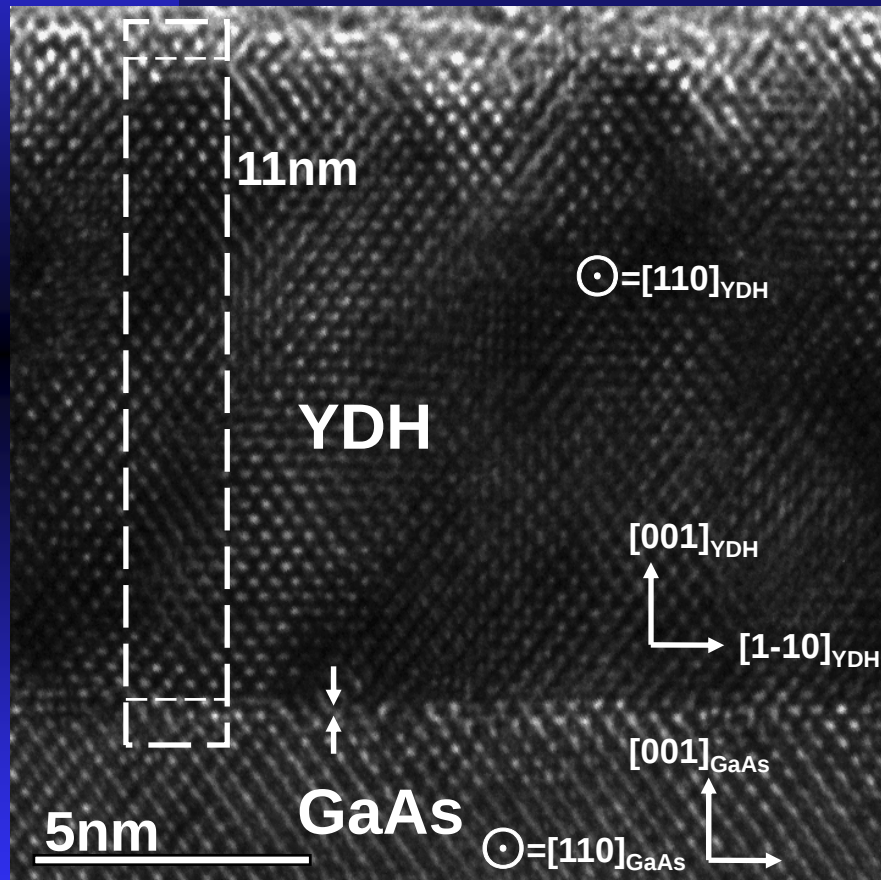


Increase of κ from 15 to over 30 in cubic phase !

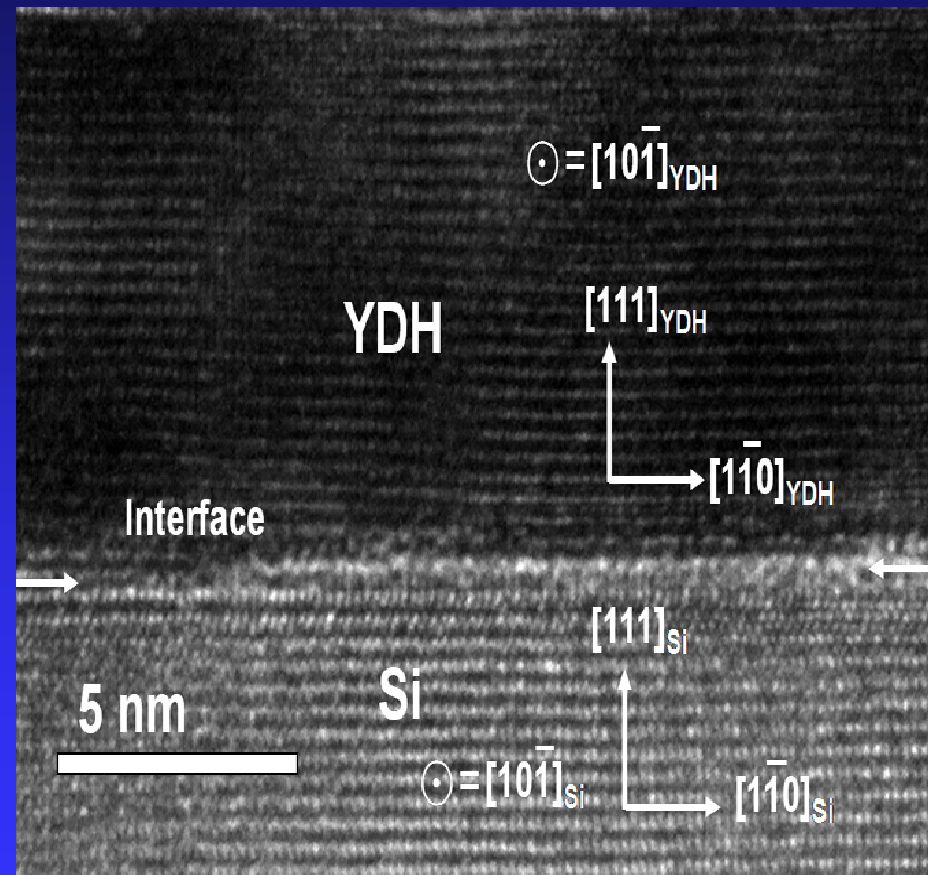
Cross Sectional HRTEM Study of The Y-doped HfO_2 Films in Cubic Phase



Interfaces of YDH(100)/GaAs, and YDH(111)/Si are atomically sharp

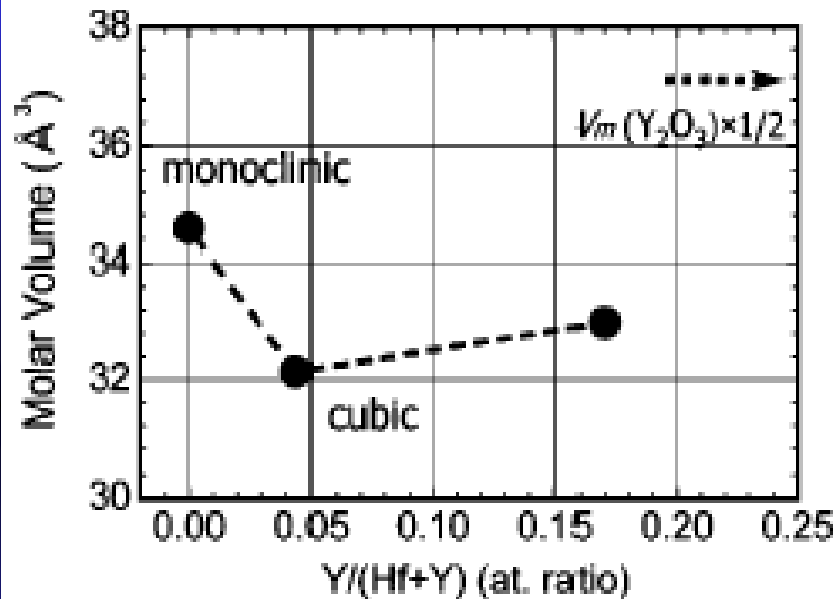


HRTEM image of yttrium-doped HfO_2 films 11 nm thick on GaAs (001).



HRTEM image of yttrium-doped HfO_2 films 7.5 nm thick on Si (111).

The Enhancement of κ through “Phase Transition Engineering”



$$\kappa = (1 + 8\pi \alpha_m / 3V_m) / (1 - 4\pi \alpha_m / 3V_m)$$

Clausius-Mossotti Relation

Change of molar volume in Y doped HfO₂

- Many high κ materials, such as HfO₂, ZrO₂, TiO₂, Ta₂O₅, commonly have high temperature phases with a higher κ .
- Achieve the enhancement through **phase transition engineering** by additions of dopants such as lower valence cations, followed by proper post high temperature anneals.

*IETS Study to Detect
Phonons and Defects in High κ Dielectrics*

But what is inside of high κ ?

Electrical characterization / optimization

-- Good News !!

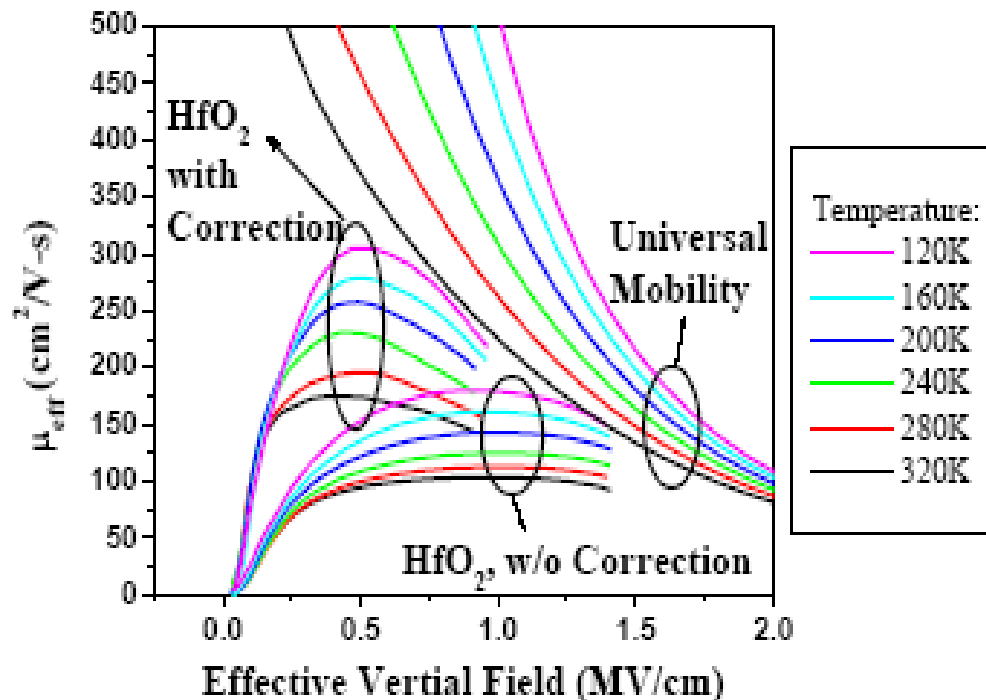
- Low electrical leakage is common .
- Have Attained an EOT under 1.0-1.4 nm .

-- Major problems are :

- High interfacial state density
- Large trapped charge
- Low channel mobility
- Electrical stability and reliability

Degradation of Mobility in High κ Gate Stack

Temperature dependence of mobility



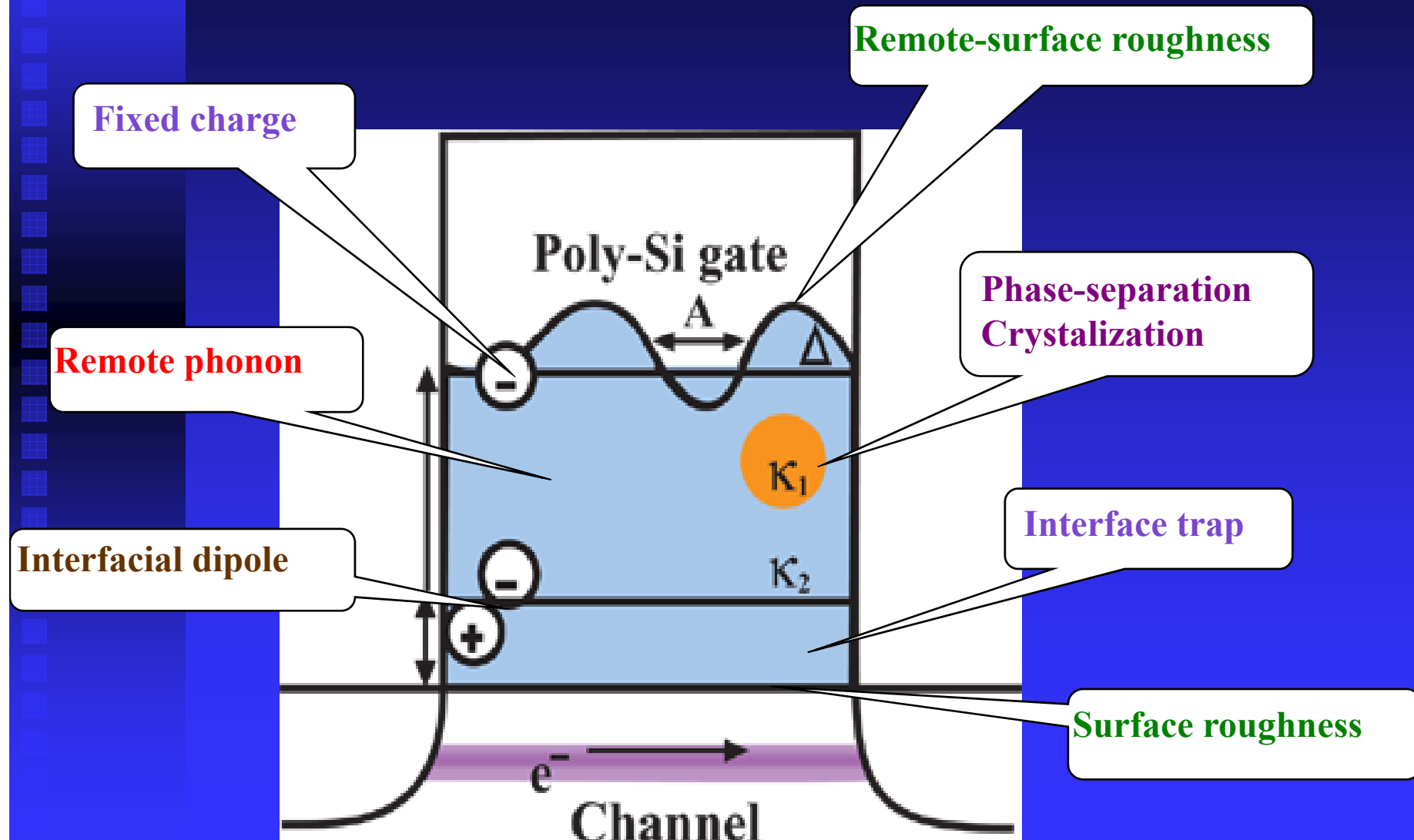
Correction from the charge trapping effect

- Effective mobility for HfO₂ is lower than universal mobility even after interface correction.

Phonons may have reduced mobility seriously !

Fechetti et al, JAP **90**, 4587, (2001).

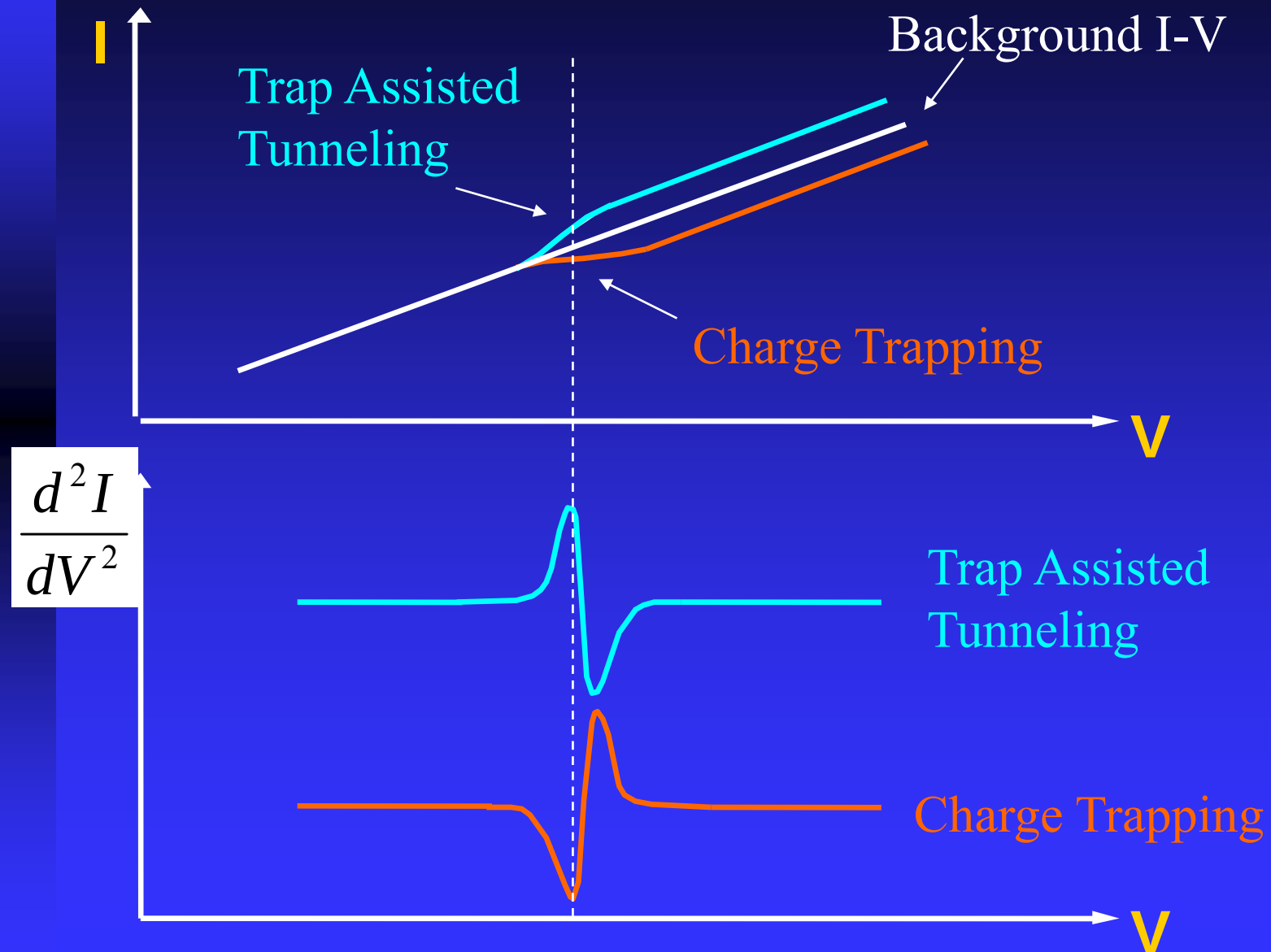
Possible Source of Mobility Degradation



Interactions Detectable by IETS

- Substrate Silicon Phonons
- Gate Electrode Phonons
- Dielectric Phonons
- Chemical Bonding
- Interfacial Structures
- Defects (Trap States)

Trap-Related Signatures in IETS

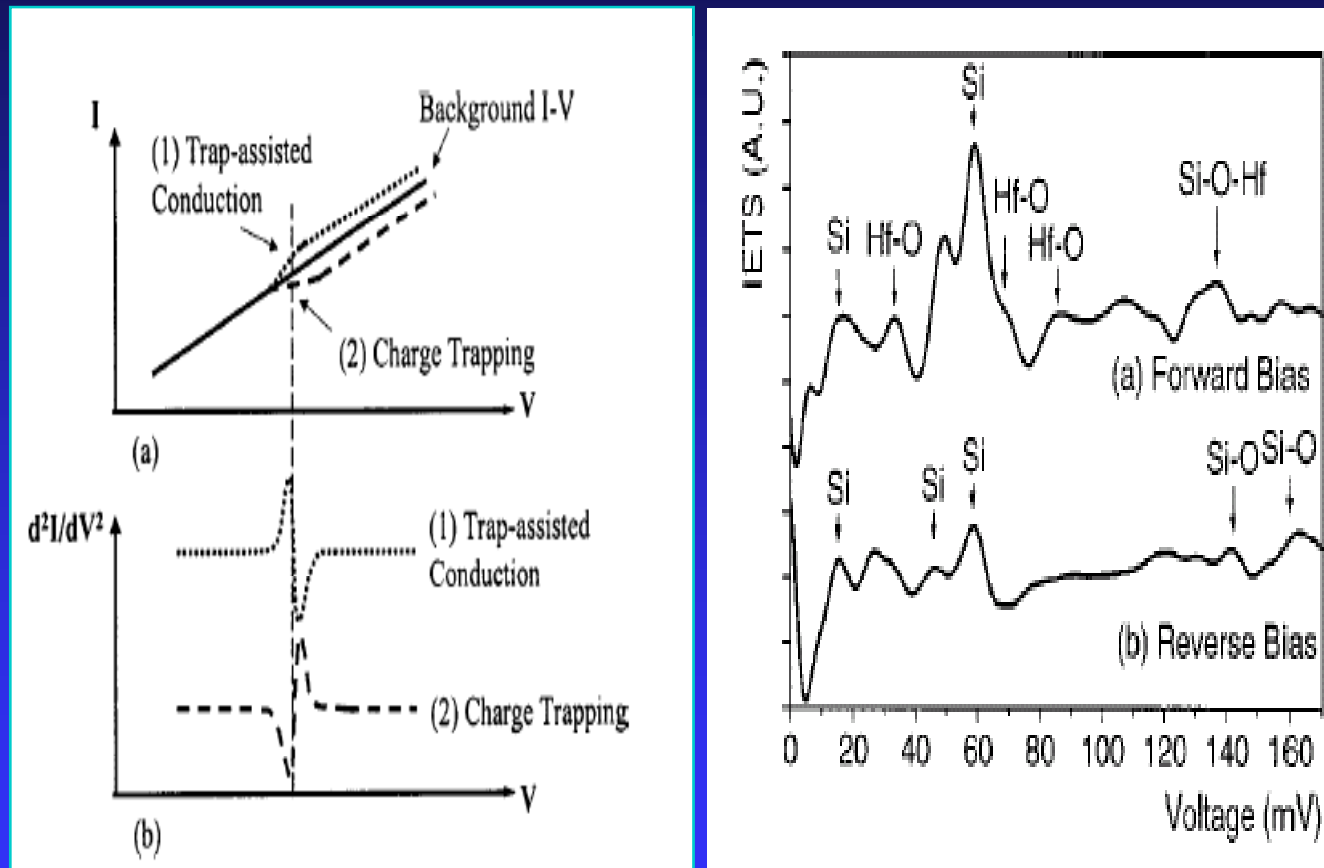


Wei He and T.P.Ma, APL **83**,2605, (2003) ; APL **83**, 5461, (2003).

Inelastic Electron Tunneling Spectroscopy (IETS)

Charge trapping will cause shift in the threshold voltage.

Trap-assisted conduction will cause increased leakage current.

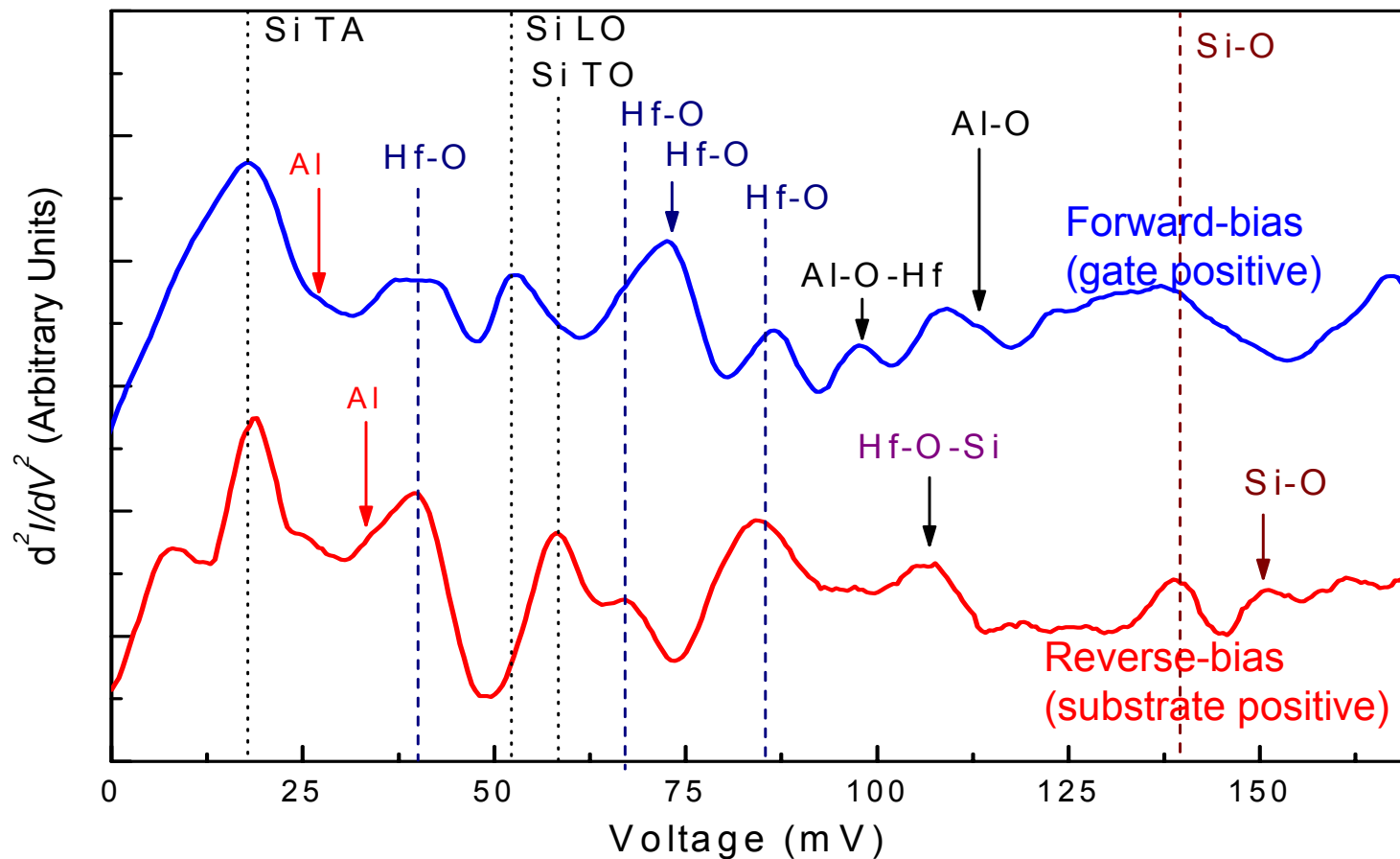


Wei He and T.P.Ma, APL **83**,2605, (2003) ; APL **83**, 5461, (2003)

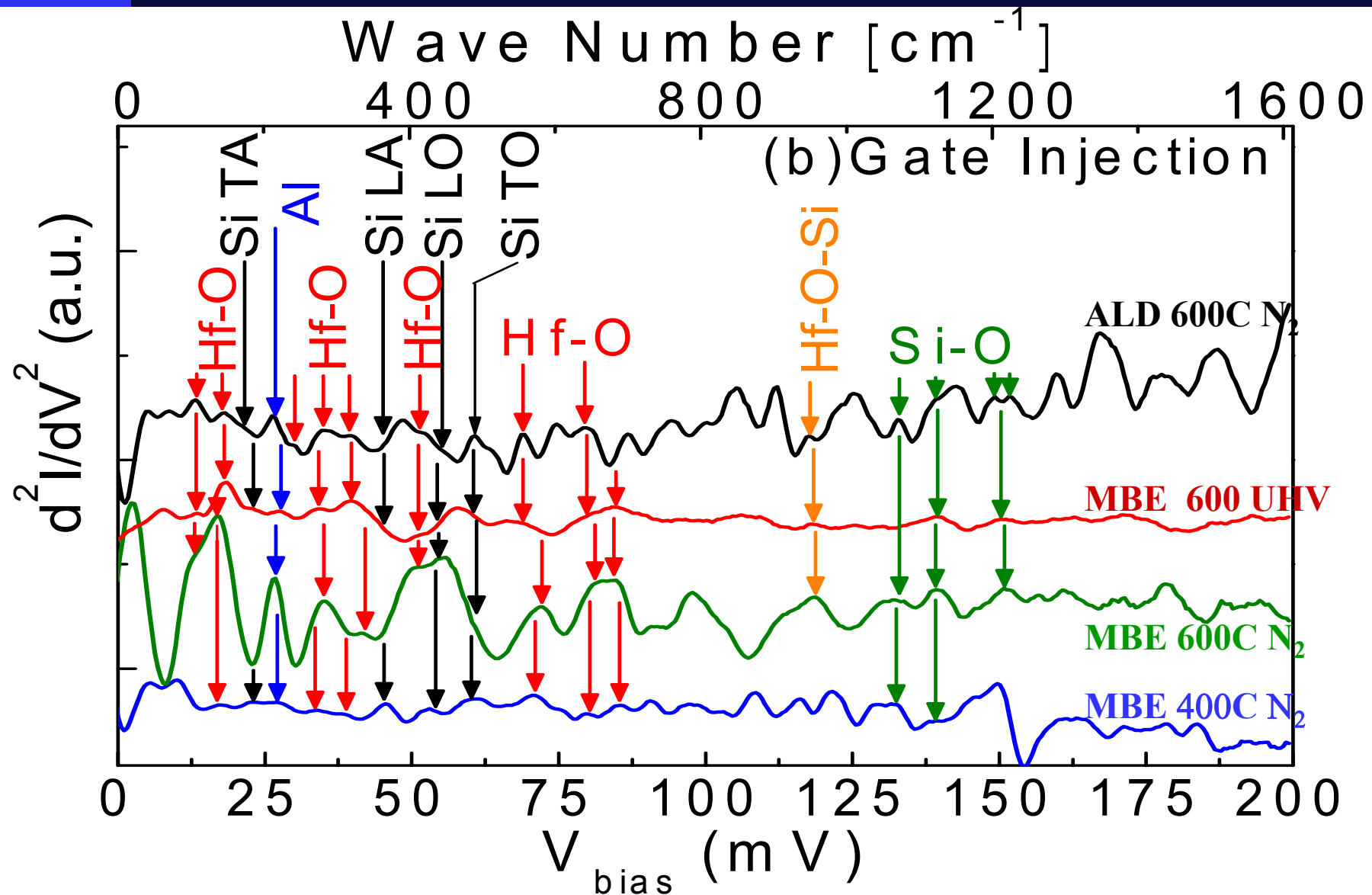
IET spectrum of Al/HfO₂/Si MOS structure

Al/HfO₂/Si, vacuum 600°C annealing for 3 minutes

d^2I/dV^2 (Arbitrary Units)

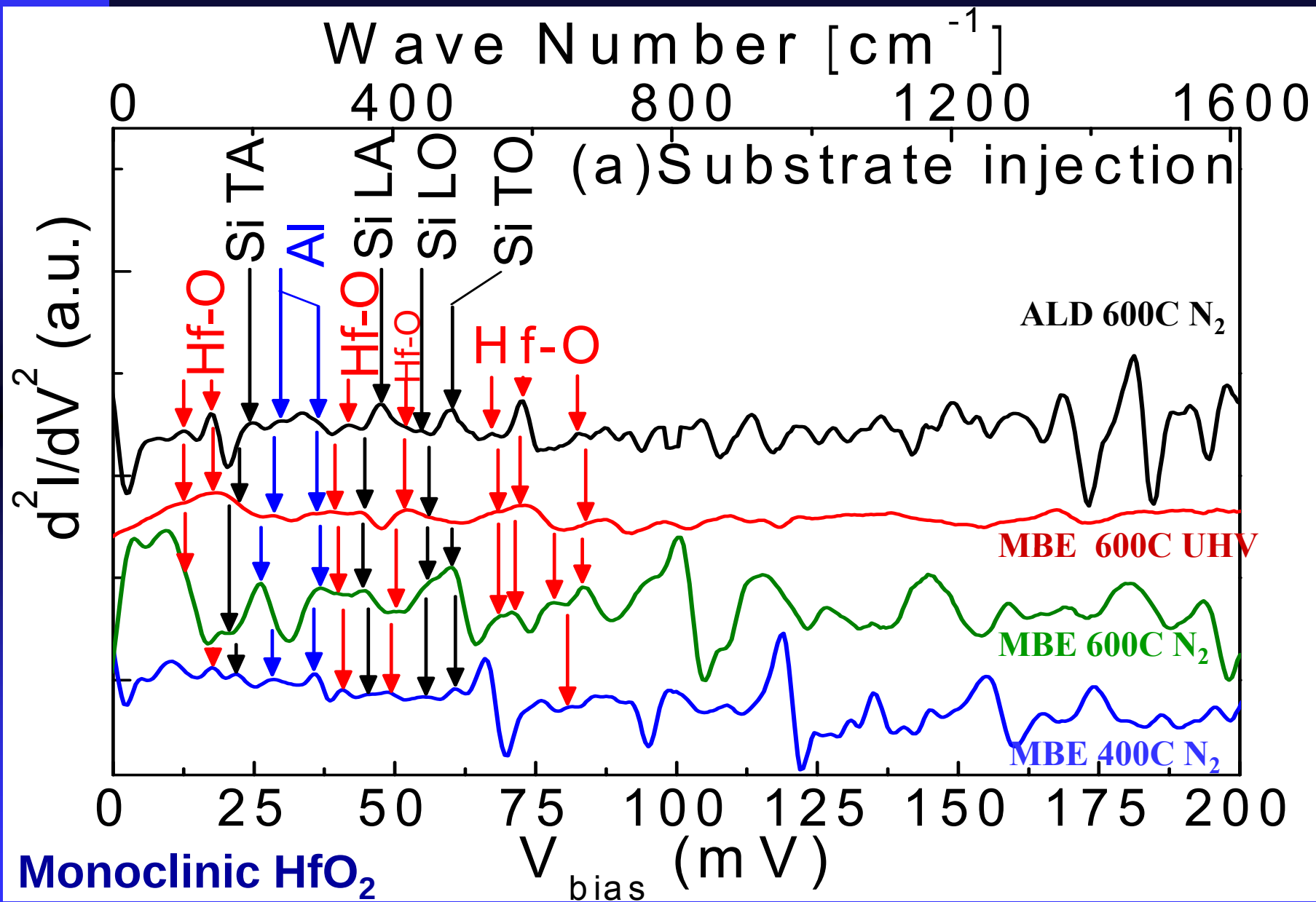


IET spectrum of Al/HfO₂/Si MOS structure

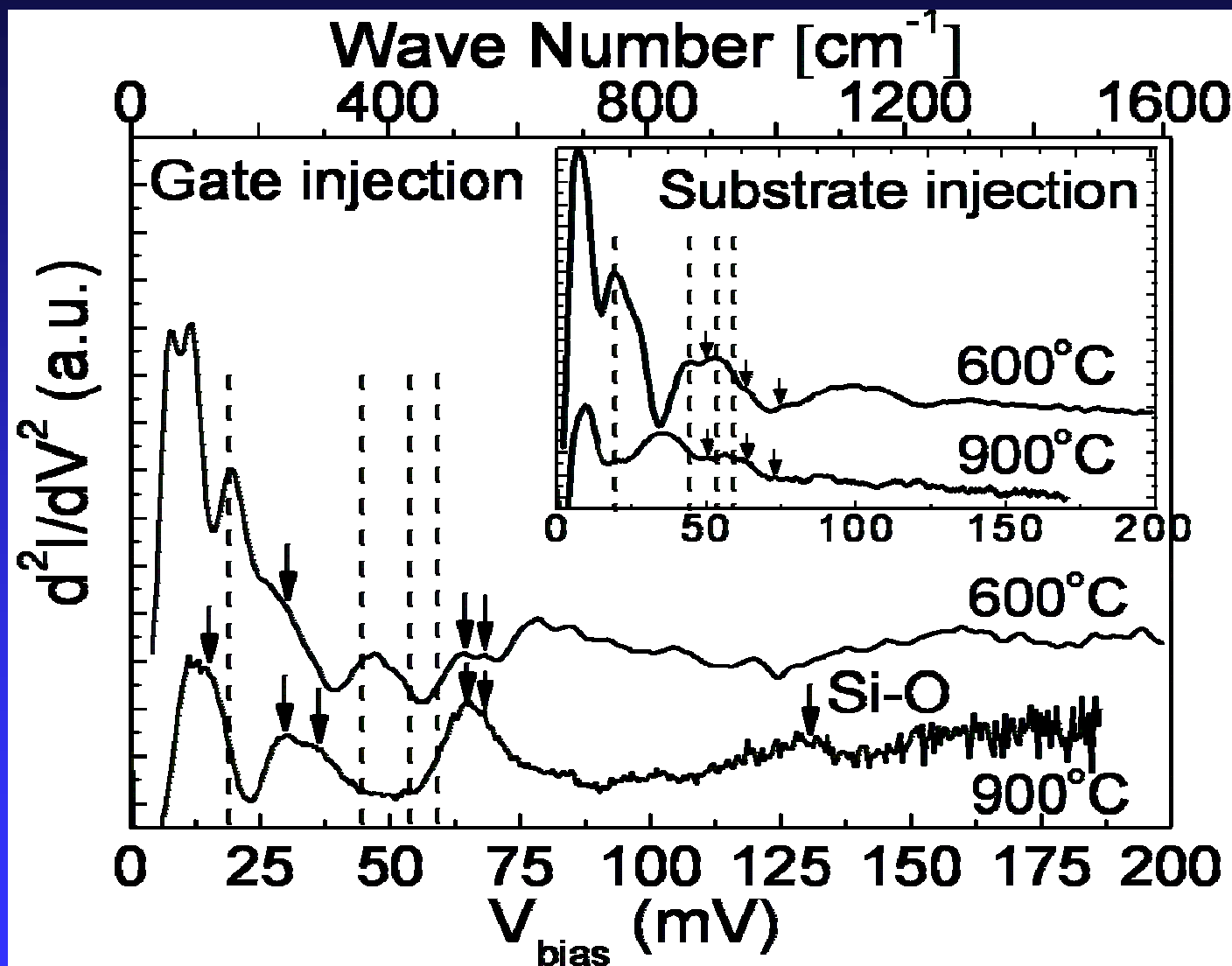


Monoclinic HfO₂

IET spectrum of Al/HfO₂/Si MOS structure



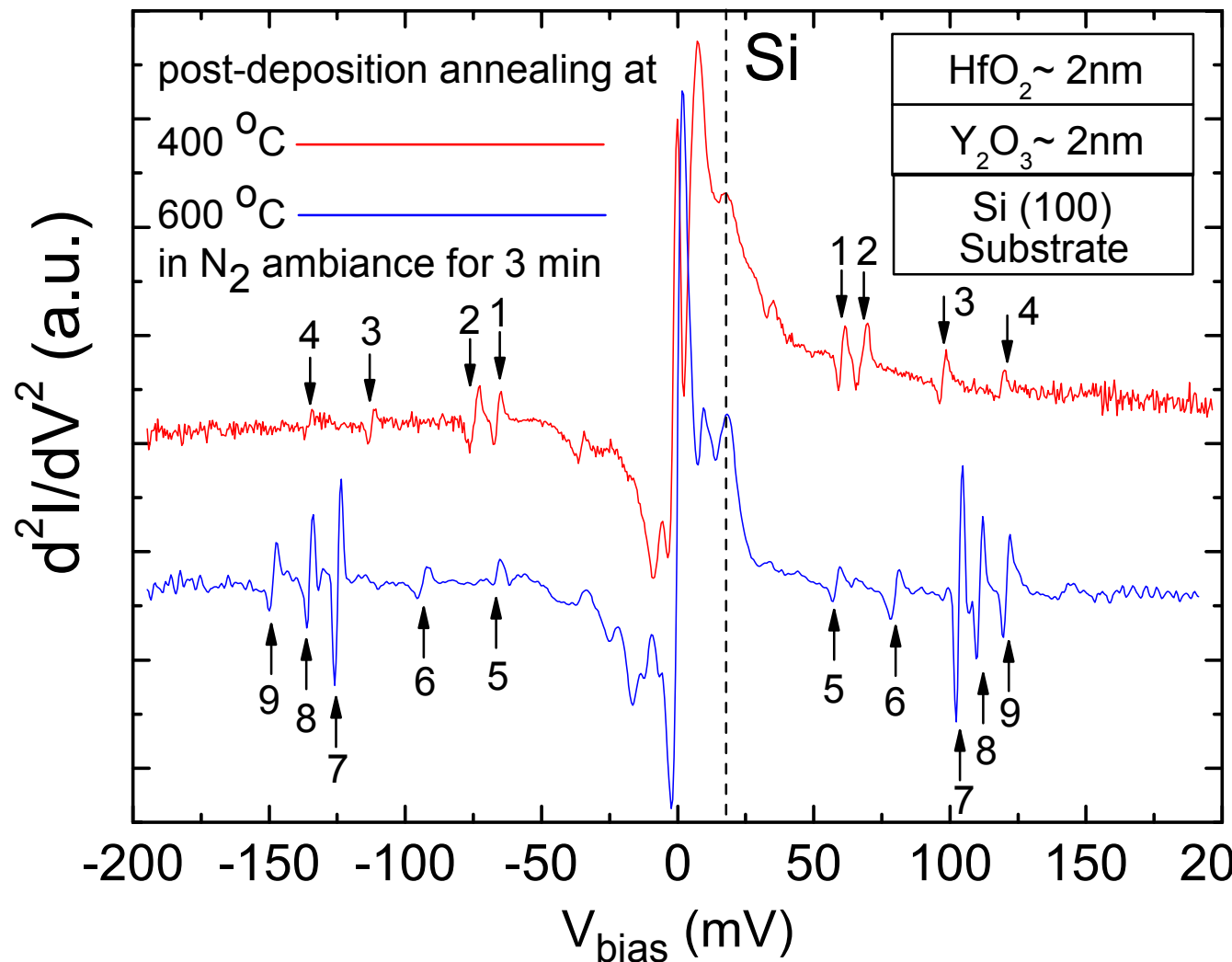
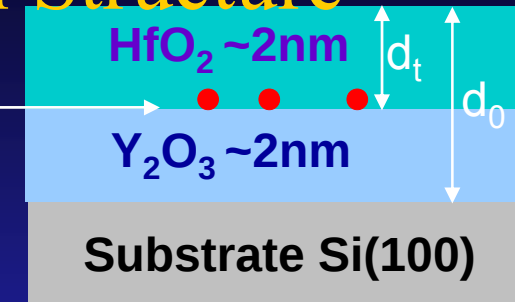
IET Spectrum of Al/Y₂O₃/Si MOS Diode



Cubic Y₂O₃

Determination of Physical Locations and Energy Levels of Trap in Stacked HfO₂/Y₂O₃/Si Structure

Charge trapping



$$V_t = V_f V_r / (V_f + V_r)$$

$$d_t = d_0 V_f / (V_f + V_r)$$

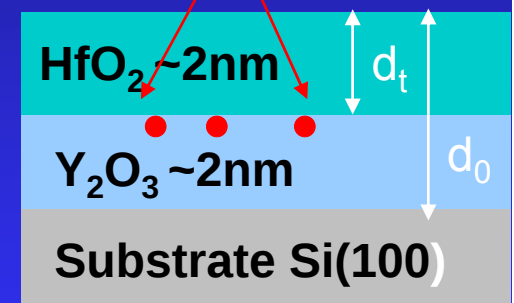
V_f and V_r are the voltages where the charge trapping features occur in forward bias and reverse bias.

M. Wang et al,
APL. 86, 192113 (2005)
APL, 90, 053502 (2007)

Determination of Physical Locations and Energy Levels of Trap in Stacked HfO₂/Y₂O₃/Si Structure

Bilayer Sample	Trap label	V_f (mV)	V_r (mV)	V_t (mV)	d_t/d_0
HfO ₂ (1.7nm) /Y ₂ O ₃ (1.4nm) 400°C	1	60	66	31	0.47
	2	67	75	35	0.47
	3	97	112	52	0.46
	4	118	135	63	0.46
HfO ₂ (1.7nm) /Y ₂ O ₃ (1.4nm) 600°C	5	58	66	31	0.46
	6	78	93	42	0.45
	7	103	124	56	0.45
	8	111	135	61	0.45
	9	120	148	66	0.45
HfO ₂ (1.2nm) /Y ₂ O ₃ (1.5nm) 600°C	1	26	32	14.3	0.45
	2	87	94	45.2	0.48
	3	99	108	51.7	0.48

Charge trapping





Can you make HfO_2 magnetic ?

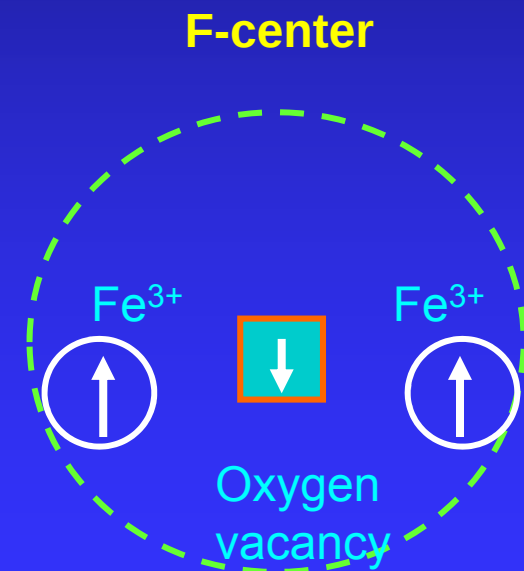
Diluted Magnetic Oxides

“Observation of Room Temperature Ferromagnetic Behavior in Cluster Free, Co doped HfO_2 Films”

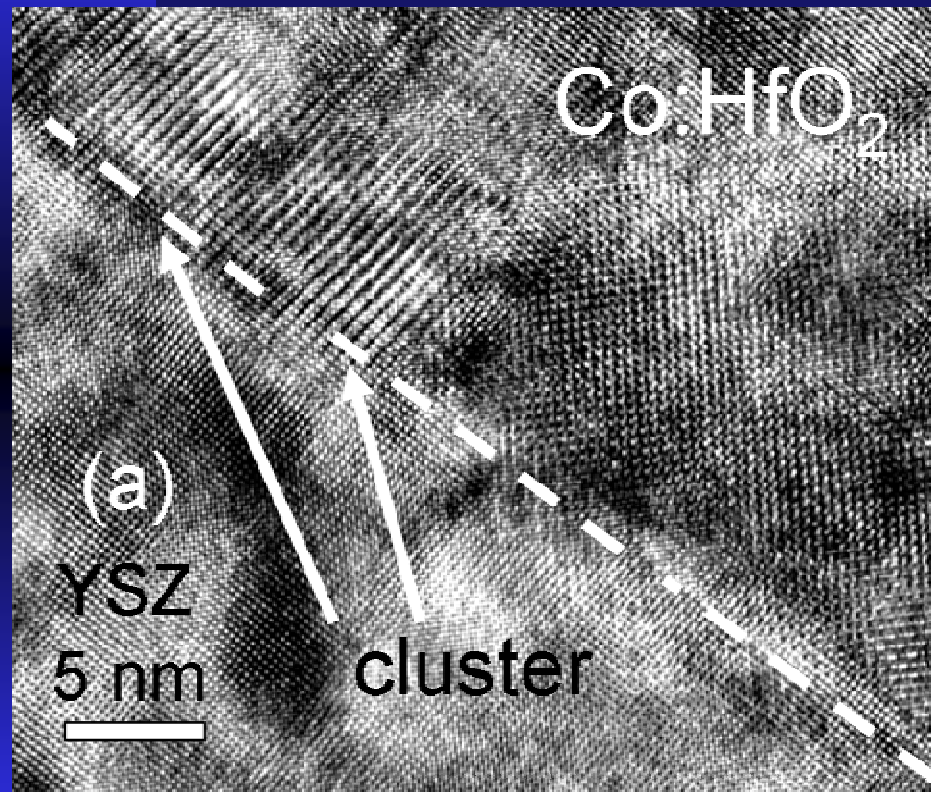
Appl. Phys. Lett. 91, 082504 (2007)

Introduction and motivation

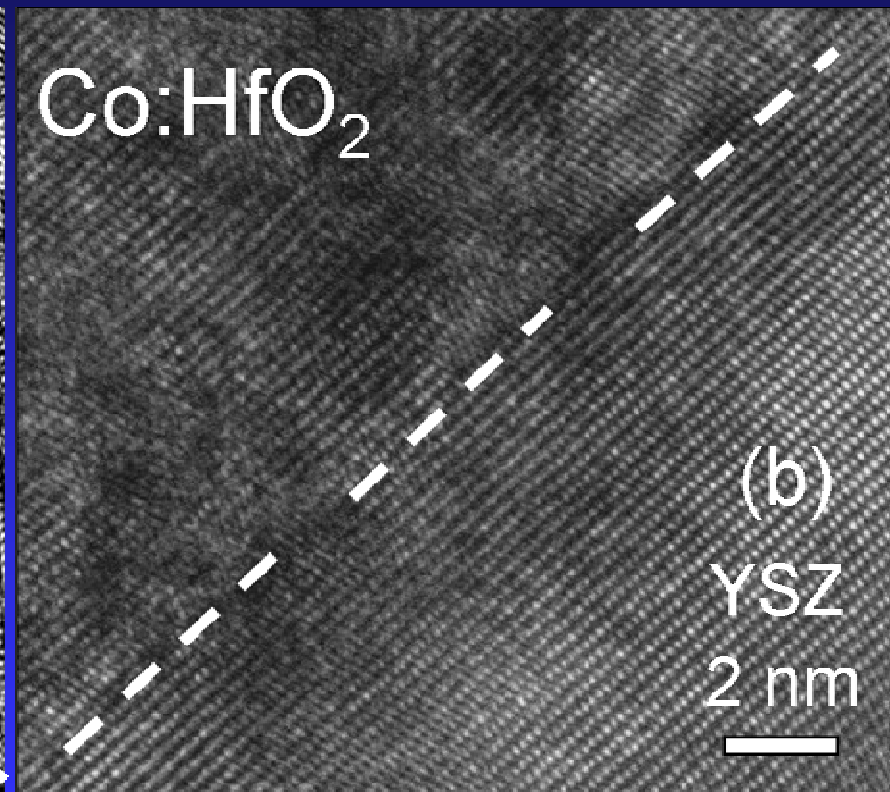
- Both injection and transport of spin-polarized carriers are necessary for the spintronic devices. Using **diluted magnetic semiconductor (DMS)** as the ferromagnetic contact is one way to achieve this goal.
- Several models such as **Zener's model**, **bound magnetic polaron**, and **F-center theory** were used to describe the ferromagnetism.
- The potential usage of **HfO₂** as alternative high- κ gate dielectrics in replacing SiO₂ for nano CMOS.
- **Giant magnetic moment** in Co doped HfO₂ as reported recently.



HR-TEM Images of High-T and Low-T Grown Films

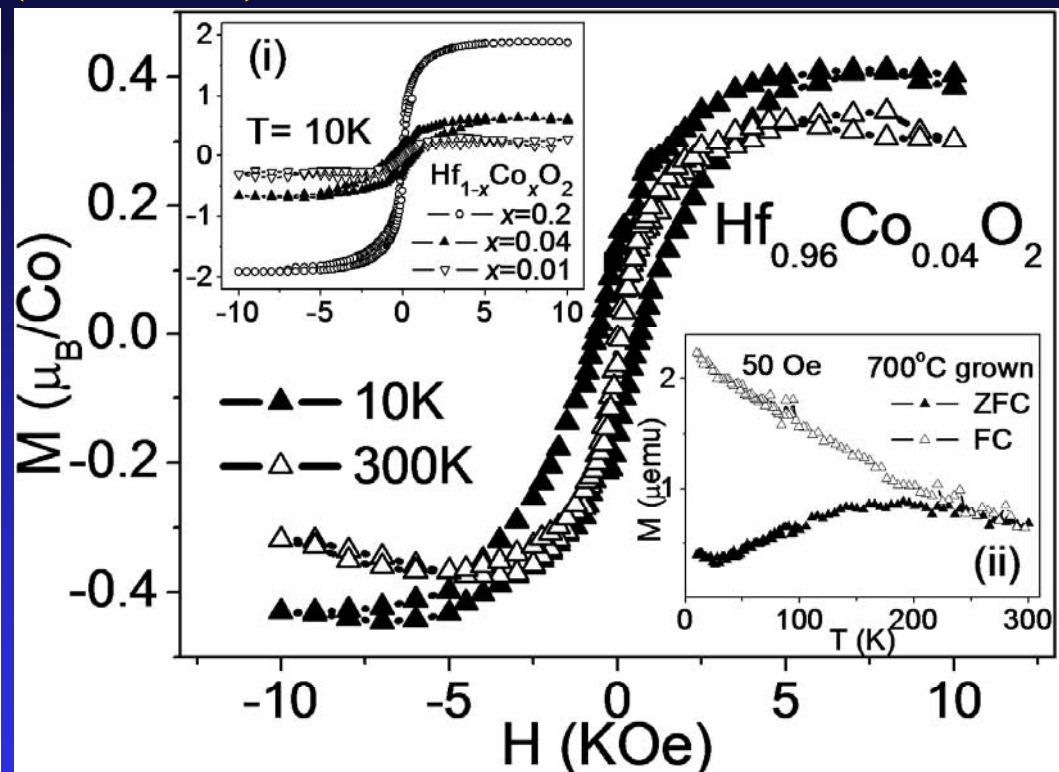
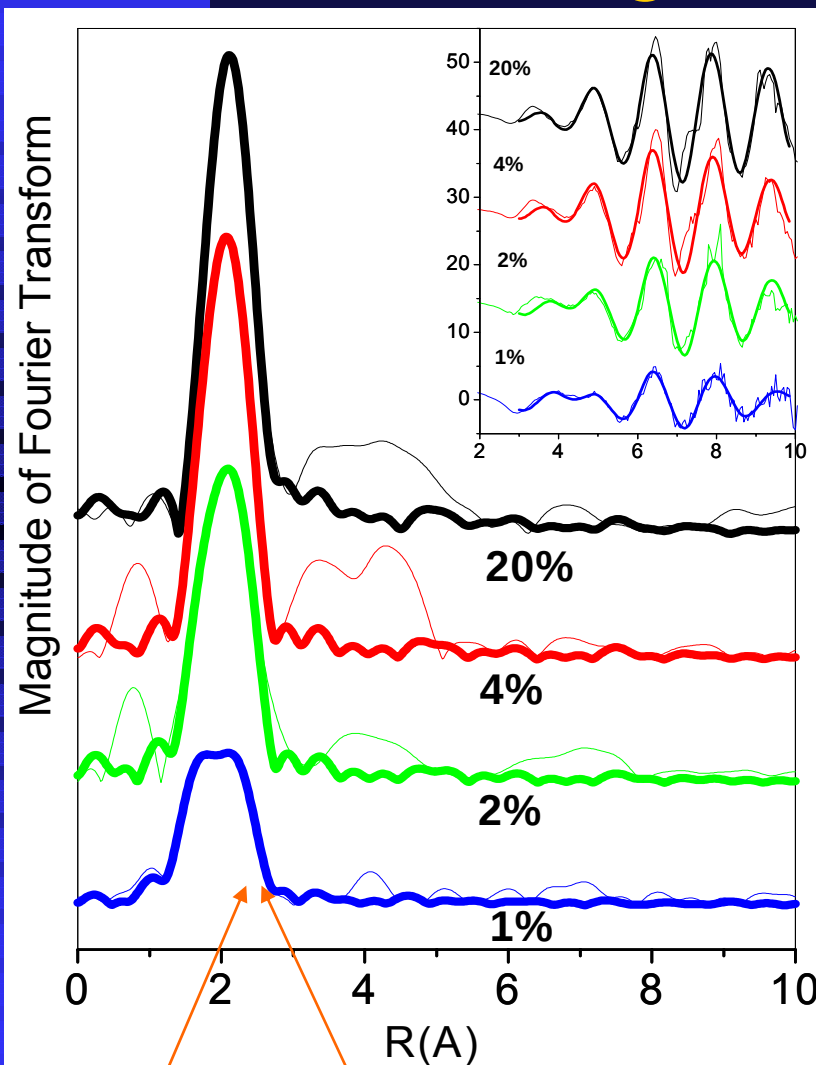


High-T (700°C) grown film
Monoclinic phase



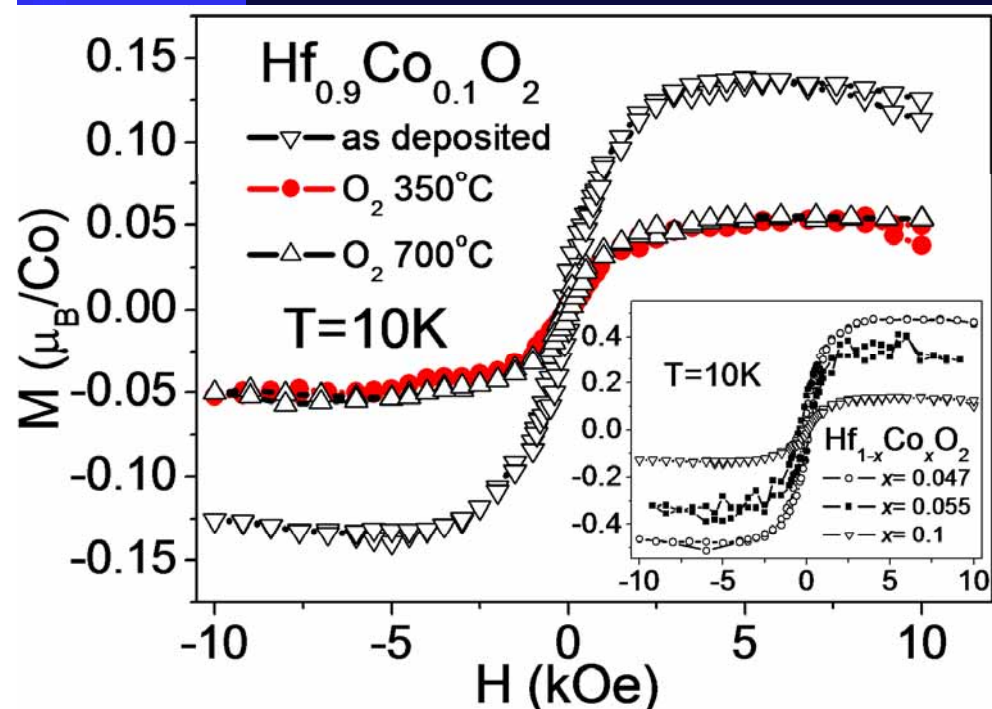
Low-T (100°C) grown,
Polycrystalline film

EXAFS and Magnetic Characterization of High-T (700°C) Grown Films



- EXAFS of the high-T grown samples showed a progressive formation of Co clusters in the film (Co = 1-20 at.%).
- Superparamagnetic temperature dependence.
- Saturation moment increases with increasing Co doping concentration.

Magnetic Property of Low-T Grown Films



Substrate	40~100	40~100	40~100
Temperature (°C)			
Doping concentration (at.%)	4.7	5.5	10
Ms at 10K (μ_B/Co)	0.47	0.36	0.13
Ms at 300K (μ_B/Co)	0.43	0.29	0.1

□ Ferromagnetic behavior was observed at both 10K and 300K.

□ The magnetic moment decreases with increasing Co doping due to enhanced dopant dopant associations.

□ The magnetic properties are stable after annealing in O₂ at 350°C.

□ Correlation between saturation magnetization with concentrations of oxygen vacancies

F-center Exchange Mechanism:

- An electron orbital created by an **oxygen vacancy** with trapped electrons is expected to correlate with magnetic spins dispersed inside the oxides.

Impurity-band Exchange Model :

- The hydrogenic orbital formed by **donor defect (like oxygen vacancy)** associated with an electron overlaps to create delocalized impurity bands.
- If the donor concentration is large enough and interacts with the magnetic cations with their 3d orbitals to form **bound magnetic polarons** leading to ferromagnetism.

$$\gamma^3 \delta_p \approx 4.3, \text{ where } \gamma = \epsilon(m/m^*)$$

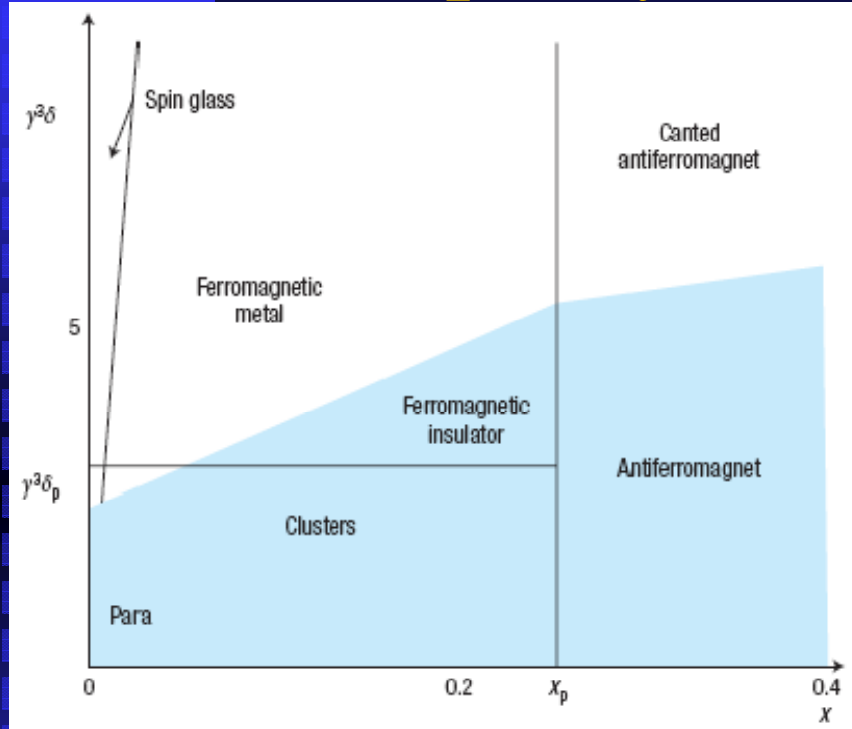
δ_p : polaron percolation threshold

X_p : cation percolation threshold

γ_H : hydrogenic radius

M. Coey et al,
Nature Materials, 2005.

Theoretical Analysis Using Impurity Band Exchange Model



Material	ϵ	m^*/m	γ	γ_H (nm)	δ_p (10^{-6})
ZnO	4	0.28	14	0.76	1500
TiO ₂	9	1	9	0.48	5900
SnO ₂	3.9	0.24	16	0.86	1000
HfO ₂	15	0.1	150	7.95	1.27
Al ₂ O ₃	9	0.23	39	2.07	72

δ_p : Polaron percolation threshold

x_p : Cation percolation threshold

γ_H : Hydrogenic orbital radius

- x_p and x_p are two landmarks on magnetic phase diagram.
- Ferromagnetism occurs when $\delta > \delta_p$ and $x < x_p$.
- δ_p of HfO₂ based DMO is about 1.27×10^{-6} - 8.15×10^{-5}
- Appearance of ferromagnetic insulator behavior in HfO₂ is more likely than ZnO, TiO₂ and SnO₂.
- Will try Y₂O₃, ZrO₂, and Al₂O₃ etc.

Major Research Topics

- Novel MBE template approach for ALD growth
- Enhancement of κ in the new phase through epitaxy
- Fundamental study by IETS for detections of phonons and defects in high κ dielectrics
- Room temperature ferromagnetism in cluster free, Co doped HfO_2 films

