

General Consideration of Scattering expt.

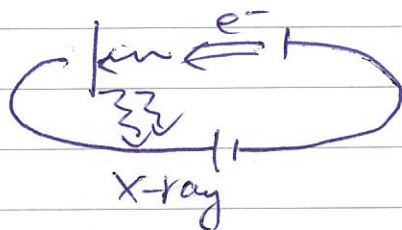
Photons, neutrons & electrons are three types of particles widely used in scattering expts. Their characteristics are different:

	X-ray	neutron	electron
charge	0	0	$-e$
mass	0	$1.67 \times 10^{-27} \text{ kg}$	$9.11 \times 10^{-31} \text{ kg}$
E (typical)	12 KeV	0.02 eV (thermal)	60 KeV
λ	$1 \text{ \AA}$	$2 \text{ \AA}$	$0.05 \text{ \AA}$
Attenuation length	100 μm	5 cm	1 μm

— The X-ray scattering is the one that is mostly common used. It only interacts with electrons. Early generation of

X-ray is by the collision of electrons

with a metal.



However, in this generation, PP% of energy is dissipated in heat.

For monochromatic X-ray, the obtained power is less than 10 W.

In the modern X-ray scattering expt, one uses X-ray generated by a synchrotron.

the power can go to 1000 W, or even more.

- neutrons : interact with nuclei & ^{weakly with} electron spins
 - Can probe magnetic structures of the materials.
 - Can be elastic or inelastic
 ⇒ hence can probe phonons.

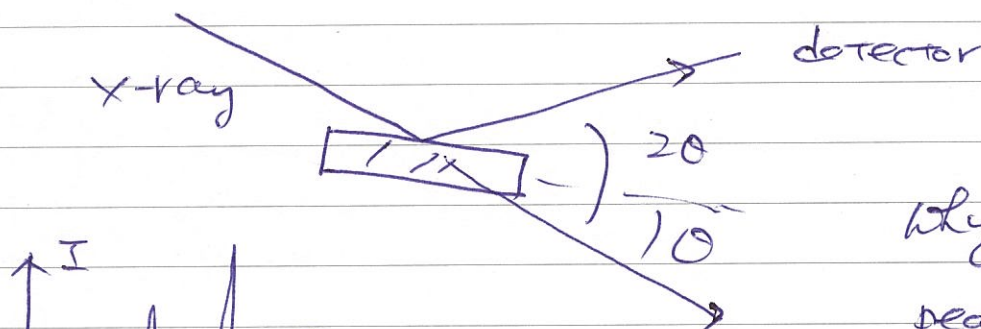
- electrons : interact strongly with matter. (electron diffraction)
 Because it's too strong, it's easily getting multiple collisions unless the sample
& damage the sample
 $\lambda \lesssim 100 \text{ \AA}$. electrons interact with

nuclei & e^- with electrostatic potentials.

Experiments are usually performed in a transmission electron microscope (TEM) or a Scanning " " (SEM) back-scattered secondary e

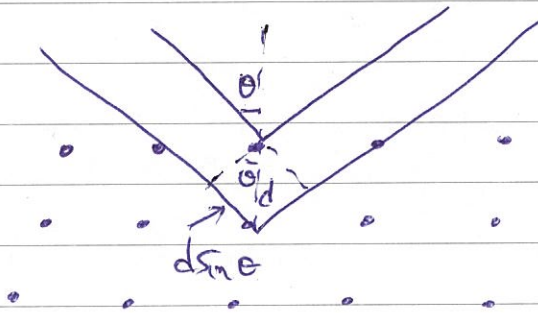
We will focus on X-ray scattering here.

X-ray scattering from crystals



Why there are peaks?
 (H.B. U.H. & W.L. Bragg)

Bragg formulation



Bragg interpreted peaks by regarding a crystal as made out of parallel planes

θ = incident angle

d = distance between planes

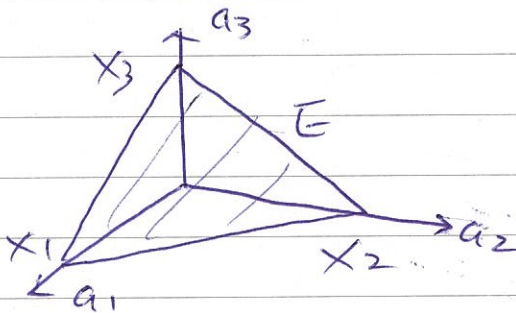
peaks result from constructive interference.

$$\therefore 2d \sin \theta = n\lambda \quad (\text{Bragg condition})$$

Characterization of lattice planes, directions:

Miller indices

(i)



$$\frac{1}{x_1} : \frac{1}{x_2} : \frac{1}{x_3} = h : k : l$$

(integers with no common factors)

E is represented by (h, k, l) (the family of planes $\{h, k, l\}$)

for negative integer use $\bar{n}$ to represent $-n$

(ii) direction $[l, m, n]$

e.g. $[100]$ $[010]$ $[001]$ $[\bar{1}00]$

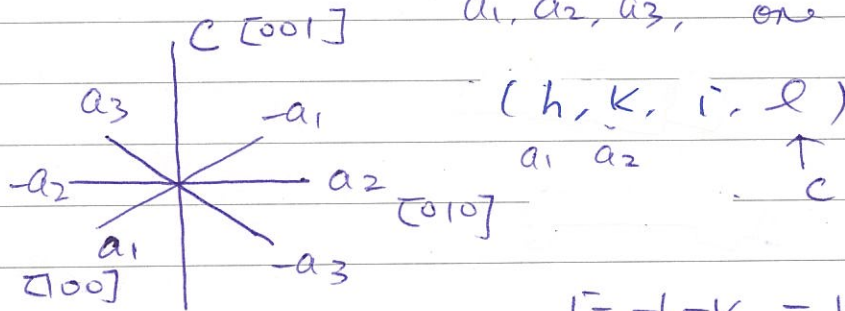
$[\bar{0}10]$ $[\bar{0}01]$ collectively $\langle 100 \rangle$

planes : (100) (010) (001) collectively as $\{100\}$.

Hexagonal systems are indexed differently:

(Bravais-Miller index)

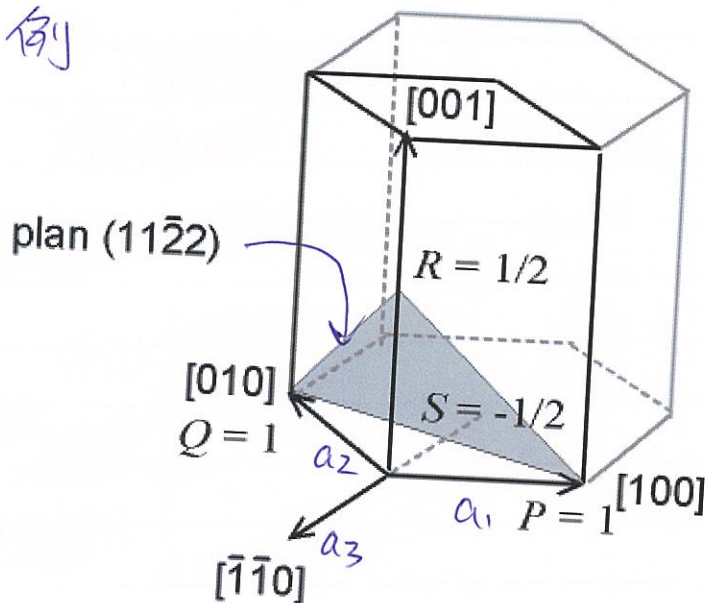
4 axes: to reflect the symmetry among a_1, a_2, a_3 , one adds one more



$i = -h - k$ = redundant index

$= \frac{1}{s}$, s = intercept of the plane with a_3

例



Indexed by this method, it would reflect symmetry easier!

Miller indices

(110) $(1\bar{2}0)$

$(\bar{2}10)$ are ~~3~~ sym.

3 planes (|| c axis)

that are symmetric

$\Rightarrow (11\bar{2}0)$ $(1\bar{2}10)$ $(\bar{2}110)$ in Bravais-Miller index exhibit the symmetry more explicitly.

Lattice planes & reciprocal lattices

Since any plane can be expressed as

$$\vec{R} \cdot \vec{r} = \text{const.} = A,$$

If one write $\vec{R}$ in terms of reciprocal primitive vectors $\vec{g}_1, \vec{g}_2$ & $\vec{g}_3$,

$$\vec{R} = l\vec{g}_1 + m\vec{g}_2 + n\vec{g}_3$$

We find that for interceptions, $\vec{r} = x_1\vec{a}_1$

$$, x_2\vec{a}_2 \text{ or } x_3\vec{a}_3, \quad \vec{R} \cdot \vec{r} = \underset{\text{Const}}{2\pi l x_1, 2\pi m x_2, \text{ or } 2\pi n x_3}$$

$$\therefore l:m:n = \frac{1}{x_1} : \frac{1}{x_2} : \frac{1}{x_3} = h:k:l$$

Given a family of parallel planes, d = distance between successive plane, we can form $e^{i\vec{R} \cdot \vec{r}}$ with

$$\vec{R} = \frac{2\pi}{d} \hat{n}, \quad \hat{n} \perp \text{planes}, \quad e^{i\vec{R} \cdot \vec{r}} = \text{const on each plane.}$$

Since $\vec{R} \cdot \vec{r} = kd = 2\pi$ for successive planes,

$e^{i\vec{R} \cdot \vec{r}}$ has the same value on all planes. Now,

one of the plane contains $\vec{r}=0$, $\therefore e^{i\vec{R} \cdot \vec{r}} = 1$ for

all planes, i.e., all $\vec{R}$! $\therefore \vec{R}$ is a

Vector in reciprocal vector space with $l, m \neq n$ being integers!

$\therefore$ Miller index $[h, k, l]$ is also the normal

vector to the plane and a vector in

reciprocal lattice!

Laue's condition & Lattice sum

The Bragg condition can be better understood by using the lattice sum.

$$I = I_0 \left| \sum_i e^{i\vec{g} \cdot \vec{r}_i} \right|^2$$

if $e^{i\vec{g} \cdot \vec{r}_i} = 1$ for all i , the I is

$$\text{maximum} \quad \& \quad I = N^2 I_0 !$$

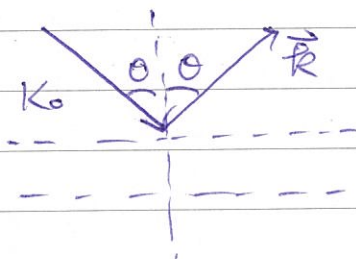
The $e^{i\vec{g} \cdot \vec{r}_i} = 1$ for all $\vec{r}_i$, implies $\vec{g} =$

reciprocal lattice vector $\vec{K}$!

i.e. $\vec{K}_0 - \vec{K} = \vec{K} = \text{reciprocal lattice vector}$

This is known as Laue's condition for peaks.

Apparently, it is equivalent to the Bragg condition in the elastic scattering: $K_0 = K = \frac{2\pi}{\lambda}$



$$2K \sin \theta = K = \frac{2\pi}{d} \times n$$

$$2d \sin \theta = n\lambda$$

Lattices with identical atoms

This is the case we have been discussed so far.

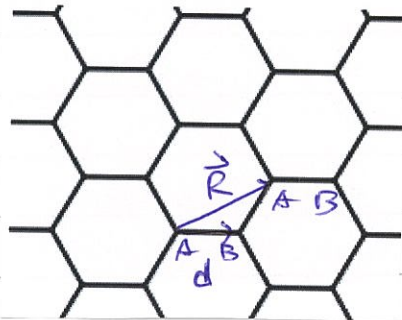
The position of each atom can
be written as

$$\vec{r}_i = \vec{R}_i + \vec{d}_j$$

$\vec{R}_i$ = a vector in the Bravais lattice,

$\vec{d}_j$ = a vector describing atoms in the
basis

example :



Then,
$$\sum_i e^{i\vec{g} \cdot \vec{r}_i} = \left(\sum_i e^{i\vec{g} \cdot \vec{R}_i} \right) \left(\sum_j e^{i\vec{g} \cdot \vec{d}_j} \right)$$

$$I = I_0 \left| \sum_i e^{i\vec{g} \cdot \vec{R}_i} \right|^2 |F_g|^2$$

F_g = geometrical structure factor.

In general, if atoms in the basis are
not identical, one includes the atomic
form factor f_j , then

$$F_g = \sum_j f_j e^{i\vec{g} \cdot \vec{d}_j}$$

f_j is essentially the scattering contribution from

atom (ion) j , it also depends on $\vec{g}$.

$$f_j = f_j(\vec{g}) = - \int d\vec{r} e^{i\vec{g} \cdot \vec{r}} \underbrace{n_j(\vec{r})}_{\text{electron density in } j\text{th atom/ion}}$$

electron density
in j th atom/ion.

It is clear that the intensity I is determined

by two factors: $\sum_i e^{i\vec{g} \cdot \vec{R}_i}$ & F_g

Both Bragg's condition & Laue's condition

only deal with $\sum_i e^{i\vec{g} \cdot \vec{R}_i}$. Therefore, they predict all peaks are equally strong!

Obviously, the introduction of basis induces

F_g so that Bragg's peaks are no longer

the same and are proportional to $|F_g|^2$!

In the extreme case, it's also possible that

$F_g = 0$ for some $\vec{g}$ and the Bragg peak completely disappear. This is known as

extinction.

* Note that a good analogy of effects due to $\sum_i e^{i\vec{g} \cdot \vec{R}_i}$ & F_g is

the interference pattern of two slits with finite width where width induces change in the peak heights. 

Example of Lattice sum $\sum_{\vec{r}} e^{i\vec{g} \cdot \vec{r}}$

1D lattice

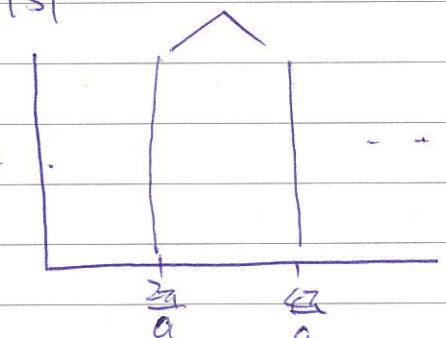
$$\sum_{\vec{r}} e^{i\vec{g} \cdot \vec{r}} \rightarrow S \equiv \sum_{n=0}^{N-1} e^{ina g}$$

$$e^{ia g} S = \sum_{n=1}^N e^{ina g}$$

$$\therefore S = \frac{1 - e^{iNa g}}{1 - e^{ia g}} \quad |S|^2 = \frac{\sin^2 \frac{Na g}{2}}{\sin^2 \frac{a g}{2}}$$

$$\frac{a g}{2} = h\pi \quad |S|^2 = N^2 \quad O(N^2)!$$

↑
enhancement
due to
crystallization!

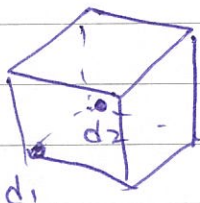


$$N \rightarrow \infty \quad \sum_n e^{ina g} \approx \sum_k N \frac{2\pi}{L} f(g - \frac{2\pi k}{a}) \quad \text{Very sharp peaks}$$

(different from peaks in liquid --)

example: BCC = Simple cubic with a basis

$$\vec{d}_1 = 0 \quad \vec{d}_2 = \left(\frac{a}{2}, \frac{a}{2}, \frac{a}{2}\right)$$



$$F_{\vec{g}} = 1 + e^{i\vec{g} \cdot \frac{a}{2}(\vec{x} + \vec{y} + \vec{z})}$$

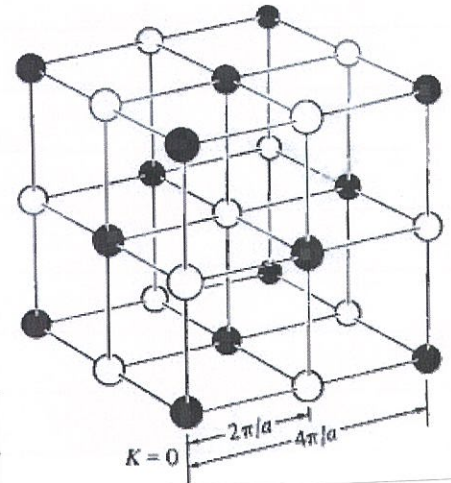
$$\vec{g} = \frac{2\pi}{a} (l\vec{x} + m\vec{y} + n\vec{z}) \quad (\text{Simple cubic})$$

$$\therefore F_{\vec{g}} = 1 + e^{i\pi(l+m+n)} = 1 + (-1)^{l+m+n}$$

$$\therefore l+m+n = \text{odd} \quad , \quad F_g = 0$$

$$= \text{even} \quad F_g = 2$$

Simple cubic (reciprocal lattice) then becomes fcc. $\rightarrow$ see right
 One can ^{also} treat the BCC as a Bravais lattice and recover the results directly.



The disappearing of peaks is due to the choice of representing the crystal as a lattice with bases.

It is therefore not extinction.

Example = diamond lattice (fcc)

$$\vec{d}_1 = (0, 0, 0) \quad , \quad \vec{d}_2 = \frac{a}{4} (1, 1, 1)$$

$$\vec{g}_1 = \frac{2\pi}{a} (-1, 1, 1) \quad , \quad \vec{g}_2 = \frac{2\pi}{a} (1, -1, 1) \quad , \quad \vec{g}_3 = \frac{2\pi}{a} (1, 1, -1) \quad (\text{bcc})$$

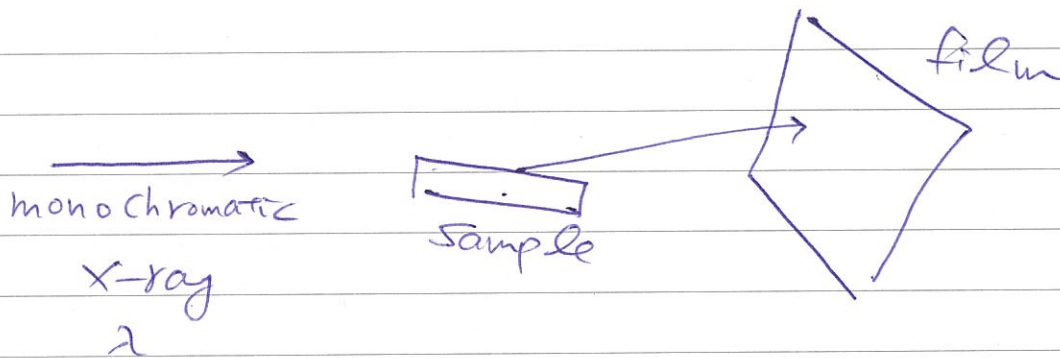
$$F_g = 1 + e^{\frac{i\pi}{2}(l+m+n)} = \begin{cases} 2 & l+m+n = 4n \\ 1 \pm i & l+m+n = \text{odd} \\ 0 & l+m+n = 2(2n+1) \end{cases}$$

extinctions $\rightarrow$

Experimental methods

From Bragg's condition, it may seem to be
& (Laue's)

straightforward to perform the expt:

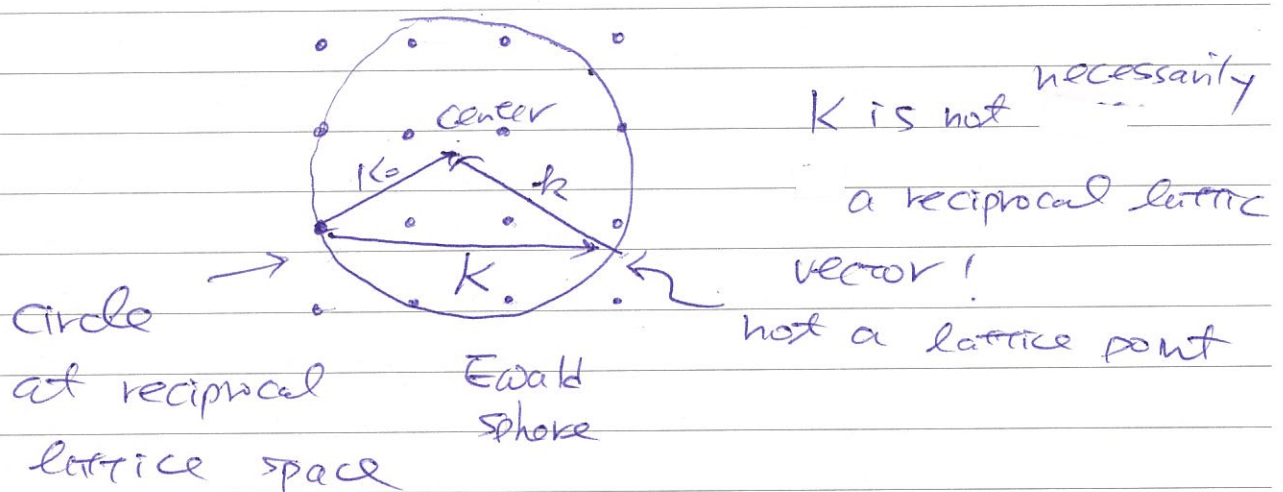


After some reflection, this clearly doesn't
work and can be summarized by the

Ewald Construction:

$$\vec{K}_0 - \vec{K} = \vec{K} = \text{reciprocal lattice vector}$$

$$K_0 = k$$



3 solutions:

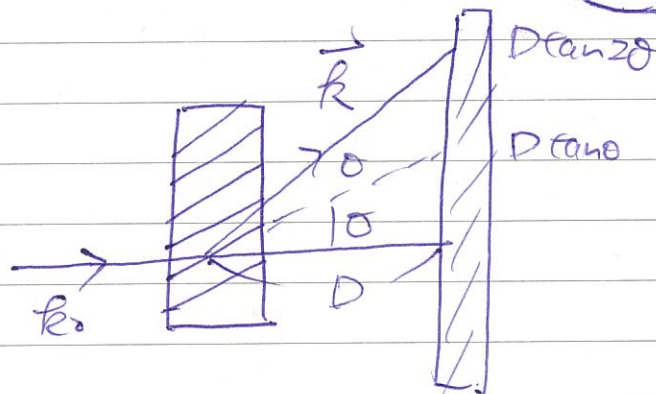
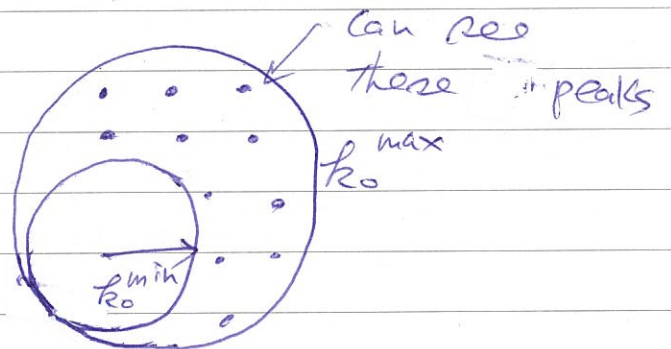
(i) Laue method

Using continuous spectrum of X-ray



Bremsstrahlung
with tungsten anode.

then, $k_0^{\min} \leq k_0 \leq k_0^{\max}$



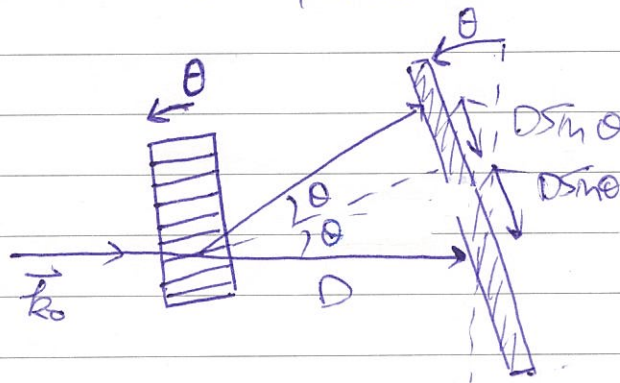
disadvantage : there are intensity variation for I_0 , it would then be difficult to pin down the intensity changes due to $|k_g|^2$

but this is a good way to orient the sample with known lattice parameters.

(ii) rotating crystal method

In contrast to change magnitude of $\vec{k}_0$, one can rotate $\vec{k}_0$ or equivalently rotate the sample.

Then, reciprocal lattice vectors can go across Ewald sphere and be observed

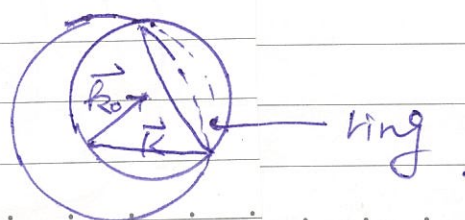


(iii) Powder Method (Debye-Scherrer Method)

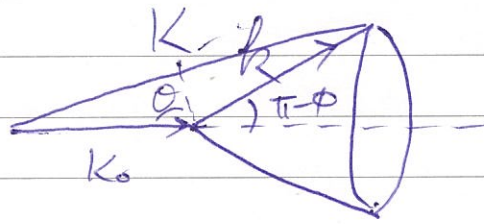
By making the sample into small crystals with all possible orientation, effectively the crystal is rotated via all angles.

Hence while $\vec{k}_0$ is fixed, reciprocal

lattice vector $\vec{K}$ is rotated and ~~is~~ becomes ^{another} sphere. Their interception is a ring:



Therefore, each K generates a cone.



$$K = 2k \sin \phi/2 \\ = 2k \sin \theta$$

+ lattice parameters (such as lattice constant $a \dots$)

Using these informations, (magnitudes of K) one can often construct the reciprocal lattice and then the real lattice, (not resolve basis)
 Bravais

In practice, ^{one} finds the crystal structure using x-ray

from synchrotron. The initial crystal orientation is usually chosen such that $\vec{k}_0 \perp$ one of facets in the sample.
 well-formed

From the preliminary data, one tries to determine allowed symmetry & unit-cell dimensions. Then one determines what the following angle to be rotated.

In the process of identifying the lattice symmetry (cubic, tetragonal, orthorhombic, ...), it is essentially a trial-and-error process.

→ Then one uses information: density, chemical composition

determine # of atoms in each unit cell

→ find positions of atoms using relative intensities of diffraction lines.

X-ray phase problem

As mentioned, what is probed by scattering expt is the correlation function.

In x-ray scattering expt, the obtained data

are $I(\mathbf{q}) = I_0 \langle n(\mathbf{q}) n(-\mathbf{q}) \rangle$. not $n(\mathbf{q})$ itself!
 $= I_0 \langle |n(\mathbf{q})|^2 \rangle$ but $|n(\mathbf{q})|$!

If we perform Fourier transformation,

we get : $\therefore I(\mathbf{q}) = I_0 \int d\mathbf{r}_1 \int d\mathbf{r}_2 e^{i\mathbf{q} \cdot (\mathbf{r}_1 - \mathbf{r}_2)} \langle n(\mathbf{r}_1) n(\mathbf{r}_2) \rangle$
 $= I_0 \int d\mathbf{r} e^{i\mathbf{q} \cdot \mathbf{r}} \int d\mathbf{r}_1 \langle n(\mathbf{r}_1) n(\mathbf{r}_1 - \mathbf{r}) \rangle$

$$\therefore \int I(\mathbf{q}) e^{-i\mathbf{q} \cdot \mathbf{r}} \propto \int d\mathbf{r} \langle n(\mathbf{r}) n(\mathbf{r} - \mathbf{r}) \rangle$$

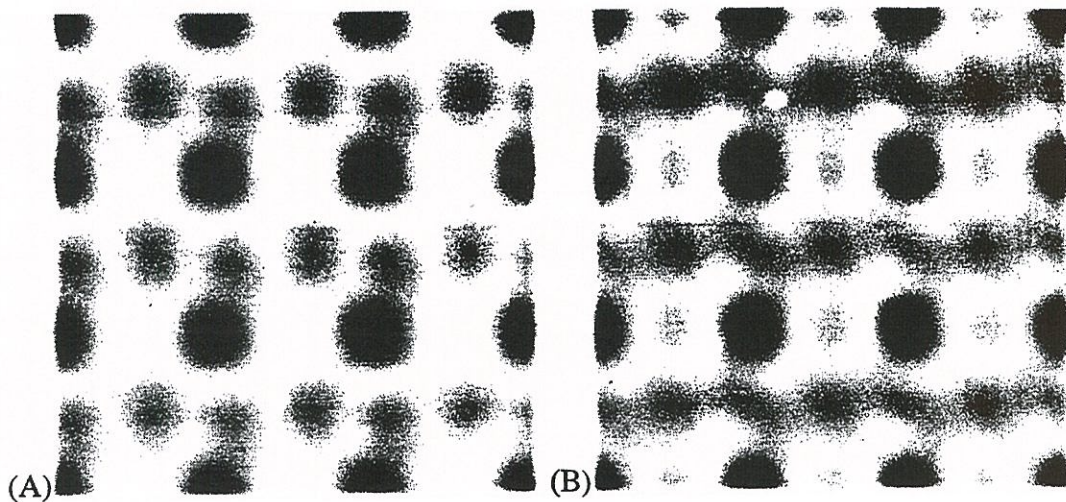
which is the correlation of density $n(\mathbf{r})$, not
directly the data of $n(\mathbf{r})$!

The problem can be traced because we measure
 $|n(\mathbf{q})|$ not $n(\mathbf{q})$ itself!

If we can determine the phase of $n(\mathbf{r})$ (i.e., measuring the electric field, not the intensity), then $n(\mathbf{r})$ is known so that $I(\mathbf{r})$ can be exactly constructed! This is known as the x-ray phase problem.

In practice, one can devise a known structure and compare $I(\mathbf{r})$ with $n(\mathbf{r})$. (see below)

They may differ a lot.



However, if one can include a heavy atom into the crystal while the crystal structure is not changed, one can use it to pin down positions of other atoms! This is usually done in determining the protein structure. Its reason can be easily understood by putting the heavy atom at $\mathbf{r} = 0$, then $I(\mathbf{r}) \sim \langle \delta n(\mathbf{r}) \delta n(\mathbf{r}-\mathbf{R}) \rangle$
 $\sim \langle n(0) n(\mathbf{r}) \rangle + \langle n(\mathbf{r}) n(0) \rangle$
 $\sim n(0) \langle n(\mathbf{r}) \rangle \propto \text{density of atoms!}$

Neutron scattering and magnetic structure

Neutrons don't carry net charge but carries spin. Therefore, their scatterings are mainly due to nuclei & magnetic structures of electrons. The scatterings from nuclei can probe phonons and we will discuss it later.

For probing magnetic structures due to spins, of electrons, we have

$$\vec{\mu}_e = -2\mu_B \vec{S}_e \quad \mu_B = \frac{e\hbar}{2m_e}$$

$$\begin{aligned} \vec{\mu}_n &= -g_n \mu_N \vec{S}_n, & \mu_N &= \frac{e\hbar}{2m_n} \\ &= -\gamma \mu_N \vec{S}_n & \vec{S}_n &= \text{Pauli matrix} \\ & & \gamma &= g_n/2 = 1.913 \end{aligned}$$

The magnetic field due to μ_e is

$$\vec{B} = \vec{\nabla} \times \vec{A}_e = \frac{\mu_0}{4\pi} \vec{\nabla} \times [\vec{\mu}_e \times \frac{1}{r}]$$

$$\vec{A}_e = \frac{\mu_0}{4\pi} \vec{\mu}_e \times \vec{\nabla} \frac{1}{r}$$

$$\therefore H = -\vec{\mu}_n \cdot \vec{B} = \frac{\mu_0}{4\pi} (2\gamma \mu_N \mu_B) \vec{S}_n \cdot [\vec{\nabla} \times (\vec{S}_e \times \frac{1}{r})]$$

$\therefore$ For a single electron spin, one has

$$I \propto |H(\vec{r})|^2 \propto \left| \frac{1}{r^2} \vec{S}_n \cdot \underbrace{\vec{r} \times (\vec{S}_e \times \vec{r})}_{\vec{S}_e^\perp} \right|^2$$

$\vec{S}_e^\perp$ (only components transverse to $\vec{r}$ contribute)

For many spins, $\vec{r} \rightarrow |\frac{1}{r-r}|$, each $\vec{S}_i$ is associated with a phase factor $e^{i\vec{g} \cdot \vec{r}_i}$

$$\therefore I \propto \left| \frac{1}{g^2} \sum_i \vec{S}_i \cdot \vec{g} \times \underbrace{\left(\sum_j \vec{S}_j e^{i\vec{g} \cdot \vec{r}_j} \right)}_{\vec{A}} \times \vec{g} \right|^2$$

$\propto \langle \vec{S}_{\frac{1}{g}} \cdot \vec{S}_{\frac{1}{g}} \rangle$ for unpolarized neutrons.

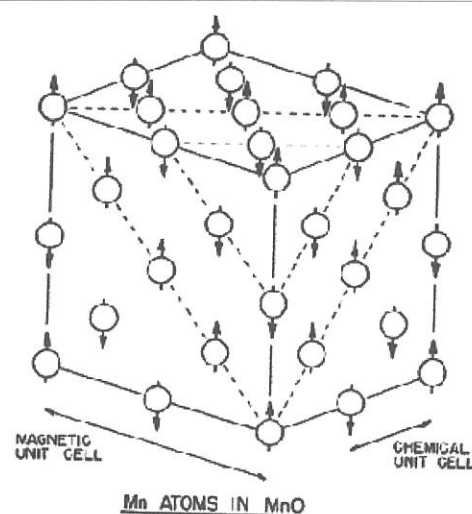
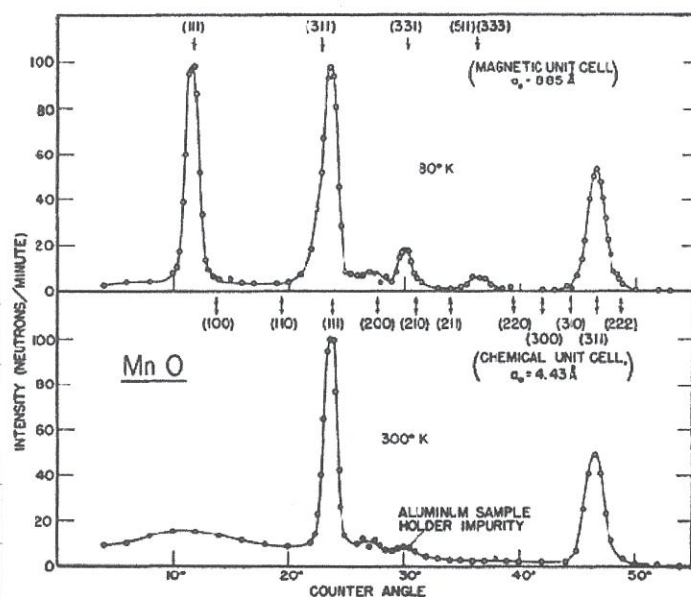
$$\langle S_n^x S_n^y \rangle = \delta_{AB}$$

$$\langle S_n^x A_x \langle B_{AB} \rangle \rangle = \langle A_x A_x \rangle$$

This is the case of elastic scattering.

Using neutron scatterings, in 1949, Clifford Shull (Phys. Rev. 76, 1256, 1964; 73, 333, 1951) first confirmed so-called antiferromagnetic structure

in MnO. In general, magnetic structures have their own symmetry groups, known as magnetic space groups.

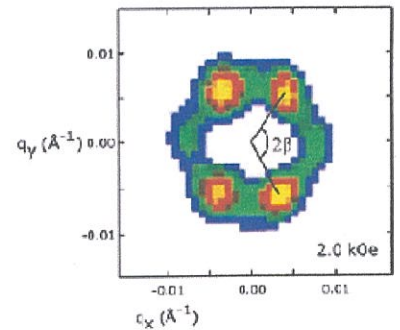
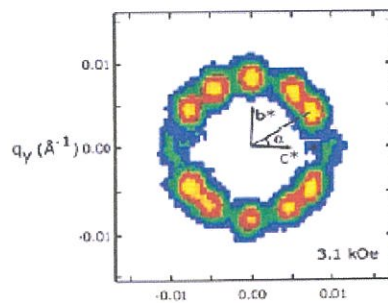
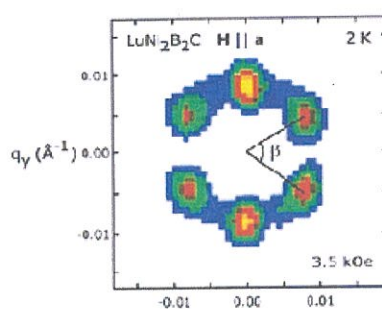
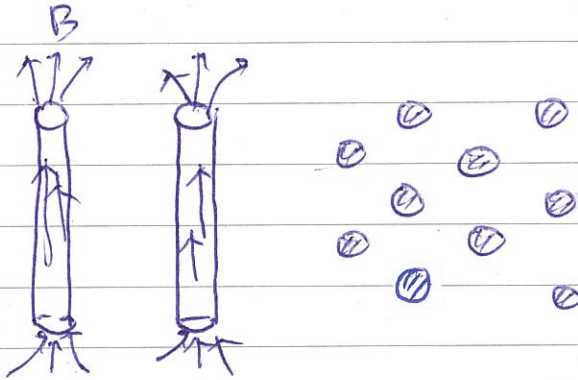


$$T_c = 120 \text{ K}$$

In addition to electron spins, neutrons can

also probe magnetic fields. For example,

vortices in superconductors form a lattice:



Small angle neutron diffraction patterns from flux lines lattice in a type-II superconductor
Note that the structure of the flux line lattice changes as a function of magnetic field
(M.R. Eskildsen et al., PRL 86, 320 (2001))

Inelastic scattering: In addition to elastic scattering,

neutrons can also excite the system and lose energy during scatterings

$$\vec{Q} = \vec{K} - \vec{K}_0, \quad \hbar\omega = E_0 - E_1$$

Clearly, in this case, one has

to replace $\vec{S}_i e^{i\vec{q}\cdot\vec{r}_i}$ by $\vec{S}_i e^{i(\omega t - \vec{q}\cdot\vec{r}_i)}$

For unpolarized neutrons,

$$I = I(\vec{q}, \omega) \propto \langle \vec{S}_i(\omega, \vec{q}) \cdot \vec{S}_i(\omega, \vec{q}) \rangle$$

Dynamical correlation function & dynamical spin susceptibility

$$\text{Since } \langle S_i^\alpha(\omega, \vec{q}) S_j^\beta(\omega, -\vec{q}) \rangle = \int dt \frac{1}{N} \sum_{\vec{r}_j} \langle e^{-i\vec{q}\cdot\vec{r}_j} S_i^\alpha(t) S_j^\beta(t) \rangle e^{i\omega t}$$

$$S_i^\alpha S_j^\beta \equiv S_{ij}^{\alpha\beta}(\omega, \vec{q})$$

What is probed in inelastic neutron expts is

the dynamical correlation function

$$\langle S_i^\alpha(t) S_j^\beta(t') \rangle.$$

The dynamical correlation function is related to the dynamical susceptibility (response function)

$$\chi \text{ defined by } \langle S_i^\alpha(t) \rangle = \int dt' \sum_{j\beta} \chi_{ij}^{\alpha\beta}(t-t') B_j^\beta(t')$$

via the so-call fluctuation-dissipation theorem

$$S_{ij}^{\alpha\beta}(\omega, \vec{q}) = \frac{1}{\pi} \frac{\text{Im } \chi_{ij}^{\alpha\beta}(\omega, \vec{q})}{1 - e^{i\omega\beta}}$$

We shall defer the detailed discussion on this theorem later.

X-ray magnetic scattering and inelastic X-ray scattering (IXS)

Light also interacts with magnetic structures via the interaction:

$$-\vec{\mu} \cdot \vec{B} = -\vec{\mu} \cdot \vec{E} \times \vec{v} = -\frac{e\hbar}{2mc} \vec{S} \cdot \vec{E} \times \frac{(\vec{p} - e\vec{A}/c)}{mc}$$

$$\rightarrow \frac{(e/c)^2}{2m^2c^2} \vec{S} \cdot \vec{A} \times \vec{A} = V_m$$

$$(\text{the other terms } -\vec{\mu} \cdot \vec{B} = -\frac{e\hbar}{2mc} \vec{S} \cdot \vec{v} \times \vec{A})$$

$$\text{Charge interaction: } \frac{1}{2m} (\vec{p} - \frac{e}{c}\vec{A})^2$$

$$\rightarrow \frac{-e}{mc} \vec{A} \cdot \vec{p} + \frac{e^2}{2mc^2} A^2 \equiv V_e$$

$$\frac{V_m}{V_e} \sim \frac{\hbar\omega}{mc^2} \sim 10^{-3} \text{ for } \hbar\omega \sim 6000 \text{ eV}$$

$$\therefore \frac{I_m}{I_e} \sim 10^{-6}$$

$\therefore$ Typical direct magnetic X-ray scattering is small. However, with high intensity

provided by synchrotron, it is still possible to do magnetic X-ray scattering. In addition

to direct scattering, resonant scattering & inelastic scattering (IXS) are also possible and being developed.